

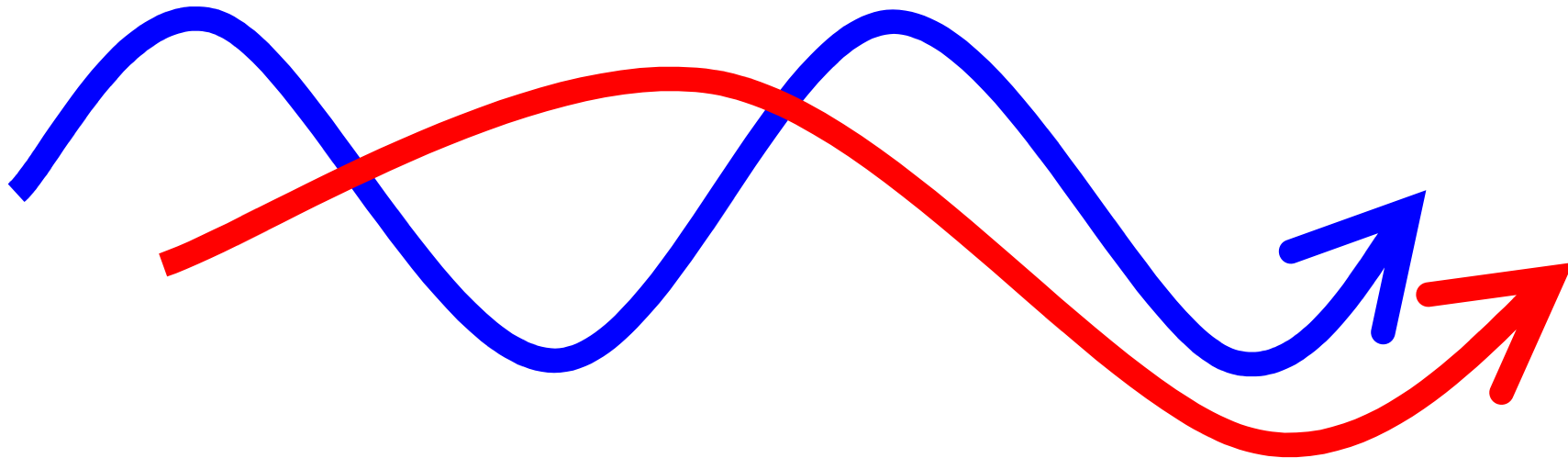
Outline of the day

- Introduction
- Basic X-ray diffraction
 - Powders, thin films, single crystals
- Small-angle X-ray scattering
- X-ray absorption spectroscopy
 - Atomic structure
 - Electronic structure (Ben Ruck)
- Advanced X-ray diffraction
 - Anomalous diffraction
 - Total X-ray scattering
- Designing experiments
 - In situ measurements

Basic X-ray Diffraction

- Peak positions
- Peak intensities
- Experiment setup
- Different types of samples
 - Single crystals
 - Powders
 - Thin films
- Information that can be obtained
 - Phase analysis
 - Crystallite size
 - Full pattern analysis
 - Preferred orientation

Properties of light

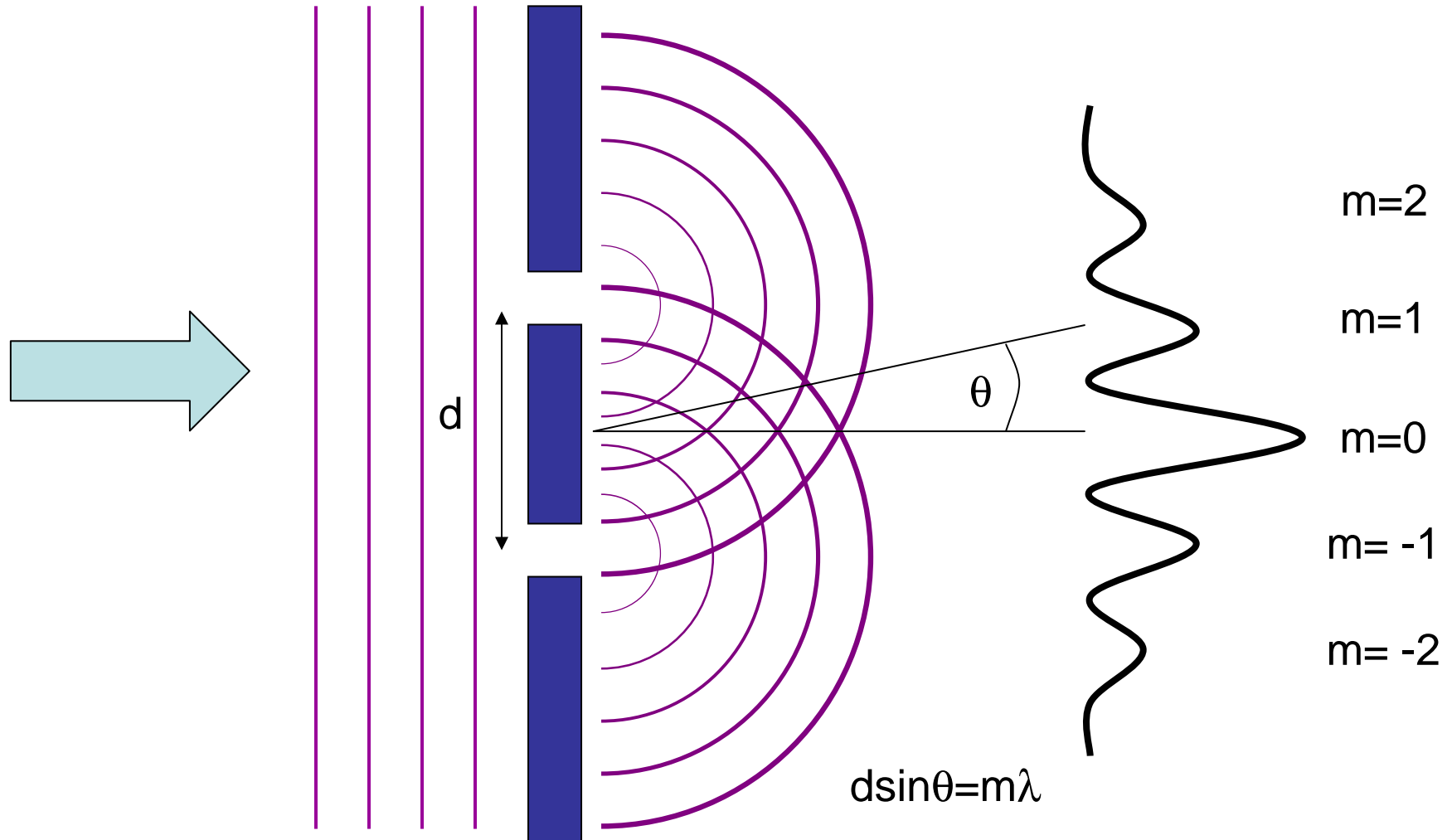


- Wavelength = $1/\text{frequency}$
- Amplitude = intensity
- Particle nature: photon

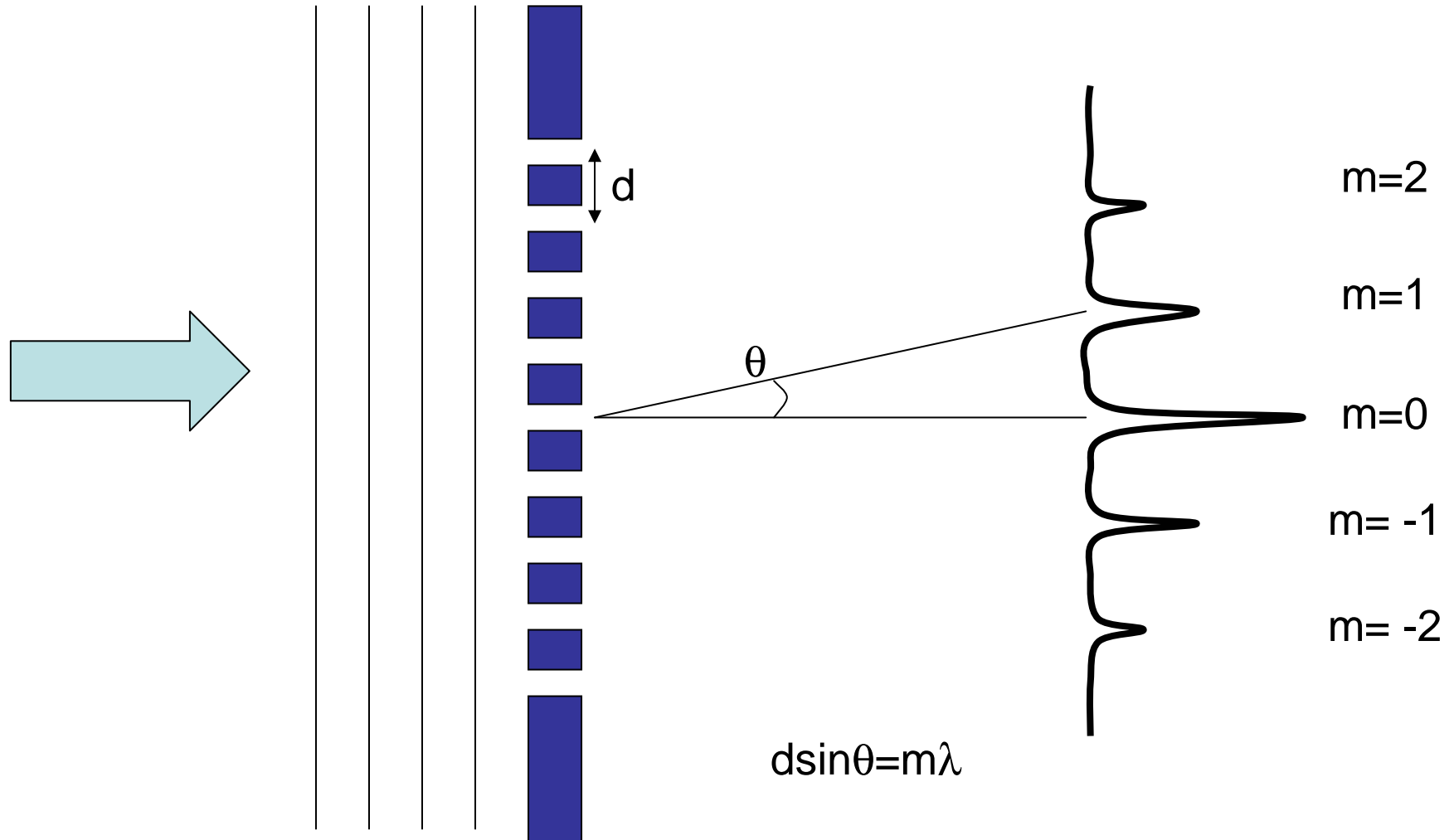
Visible
Colour

X-rays
Energy

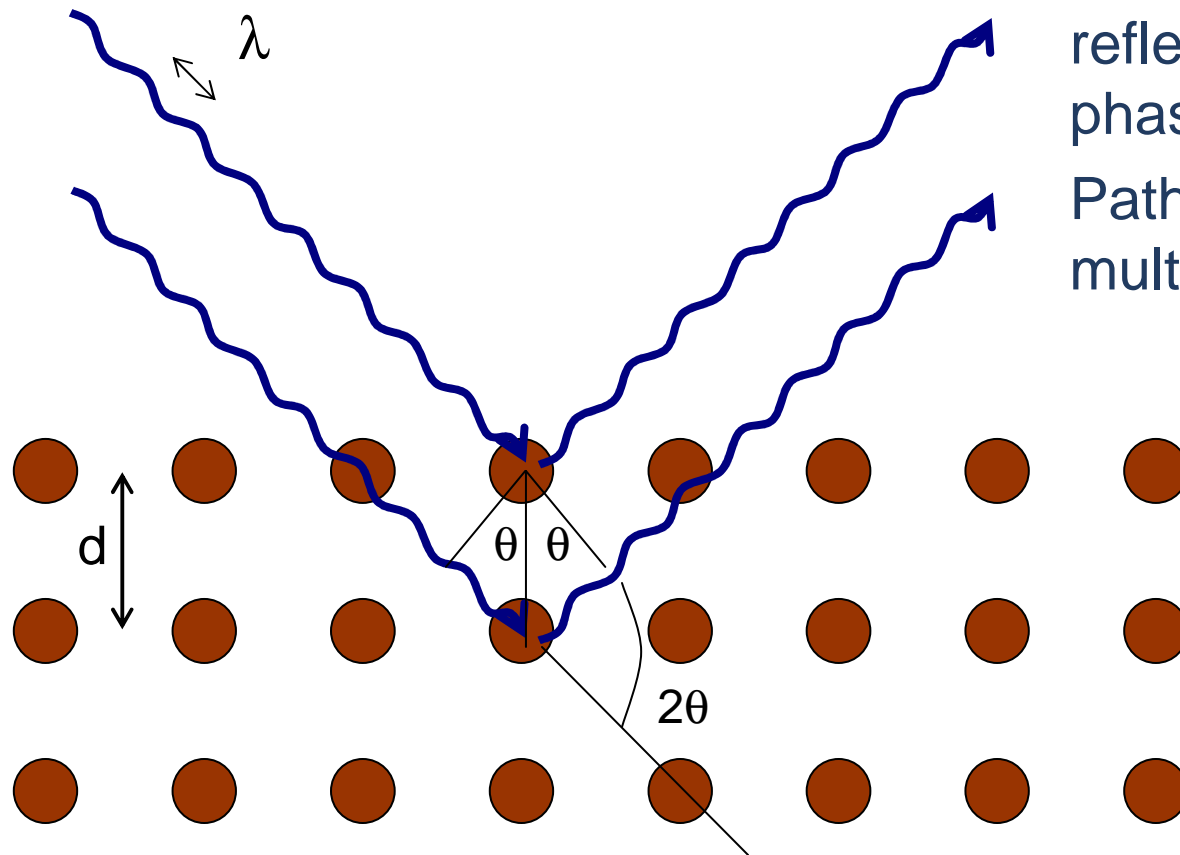
Two slit experiment



Diffraction grating



A 'grating' made of atoms

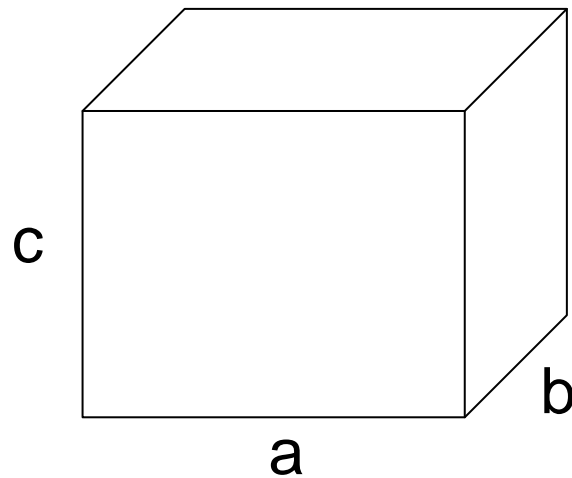
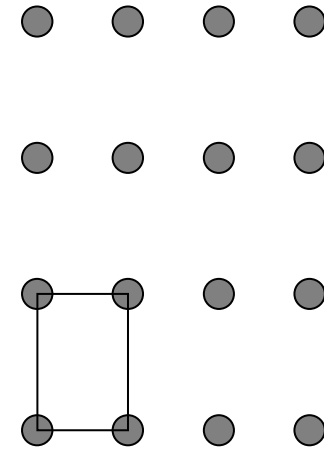
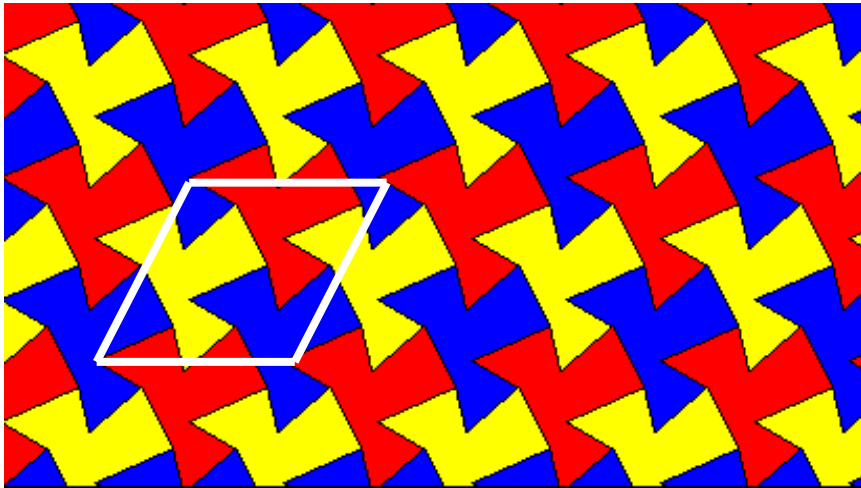


For constructive interference, reflected light must be in phase

Path difference ($2d \sin \theta$) is a multiple of the wavelength λ

Bragg's Law:
 $n\lambda = 2d \sin \theta$

The unit cell



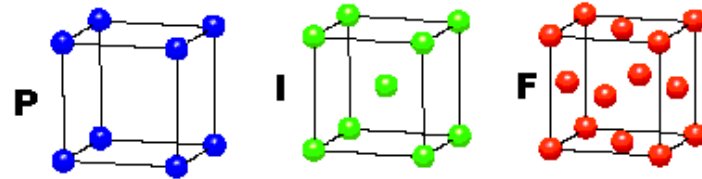
α = angle between b and c
 β = angle between a and c
 γ = angle between a and b

Bravais lattices

CUBIC

$$a = b = c$$

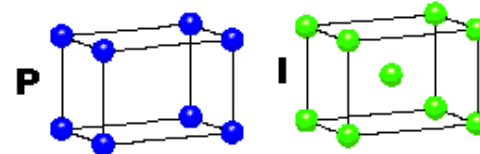
$$\alpha = \beta = \gamma = 90^\circ$$



TETRAGONAL

$$a = b \neq c$$

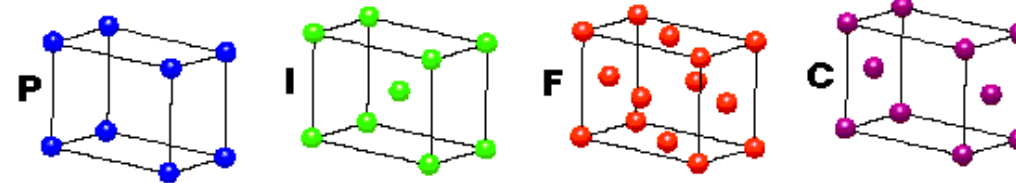
$$\alpha = \beta = \gamma = 90^\circ$$



ORTHORHOMBIC

$$a \neq b \neq c$$

$$\alpha = \beta = \gamma = 90^\circ$$

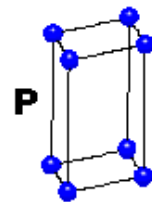


HEXAGONAL

$$a = b \neq c$$

$$\alpha = \beta = 90^\circ$$

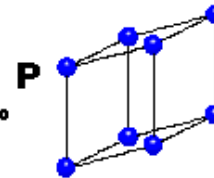
$$\gamma = 120^\circ$$



TRIGONAL

$$a = b = c$$

$$\alpha = \beta = \gamma \neq 90^\circ$$

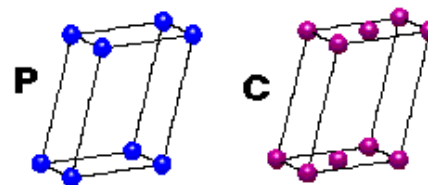


MONOCLINIC

$$a \neq b \neq c$$

$$\alpha = \gamma = 90^\circ$$

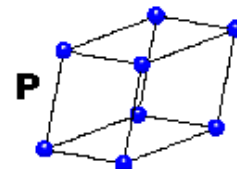
$$\beta \neq 90^\circ$$



TRICLINIC

$$a \neq b \neq c$$

$$\alpha \neq \beta \neq \gamma \neq 90^\circ$$



4 Types of Unit Cell

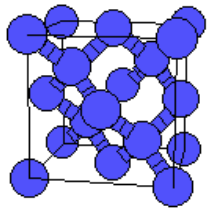
P = Primitive
I = Body-Centred
F = Face-Centred
C = Side-Centred

+

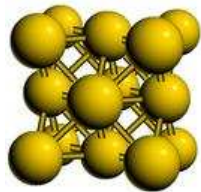
7 Crystal Classes
→ 14 Bravais Lattices

Examples

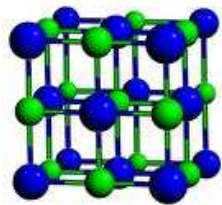
Cubic



Diamond

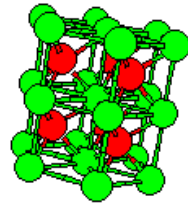


fcc

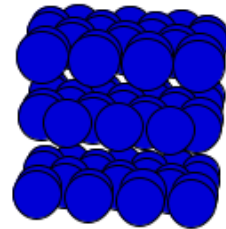


NaCl

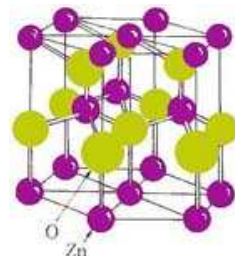
Hexagonal



WC

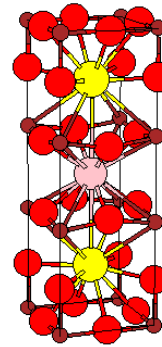


Graphite



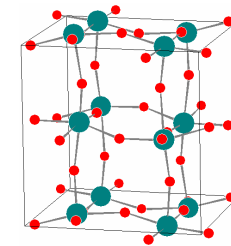
Wurtzite

Orthorhombic



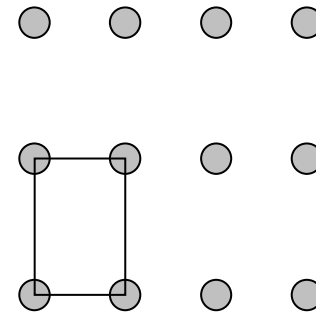
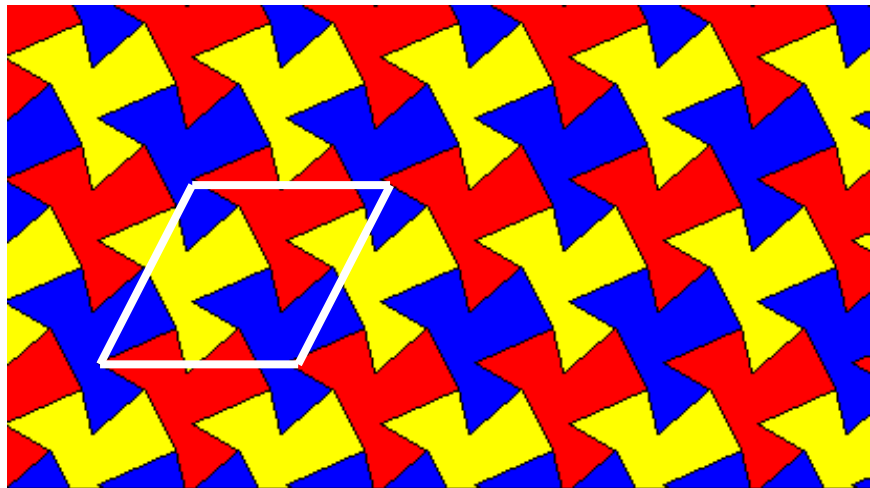
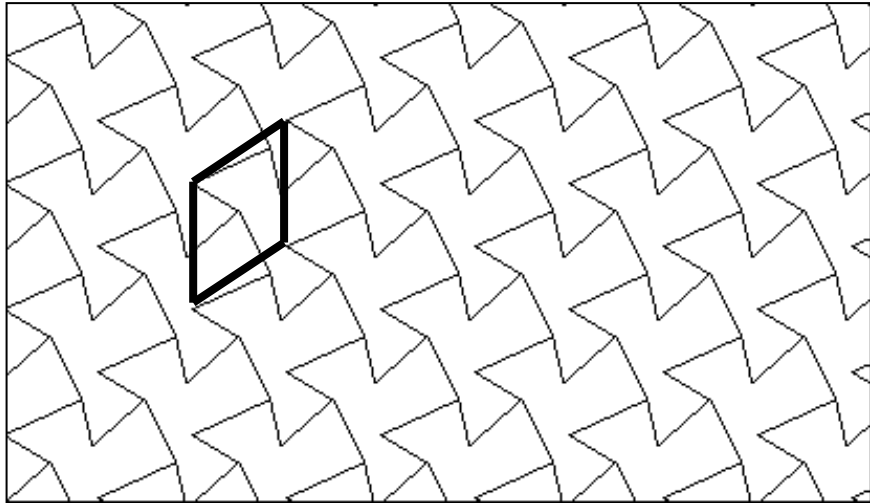
YBCO

Triclinic

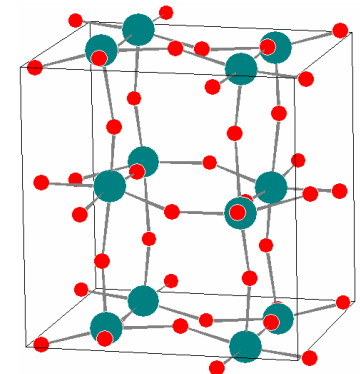
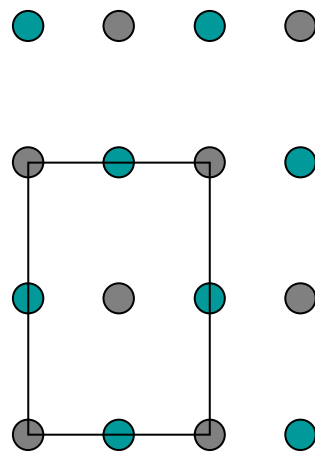


WO₃
(distorted
perovskite)

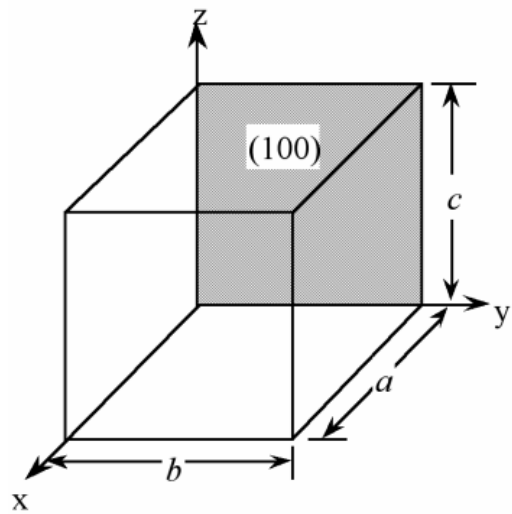
Supercells



- lattice distortions
- ordering
 - atoms
 - vacancies

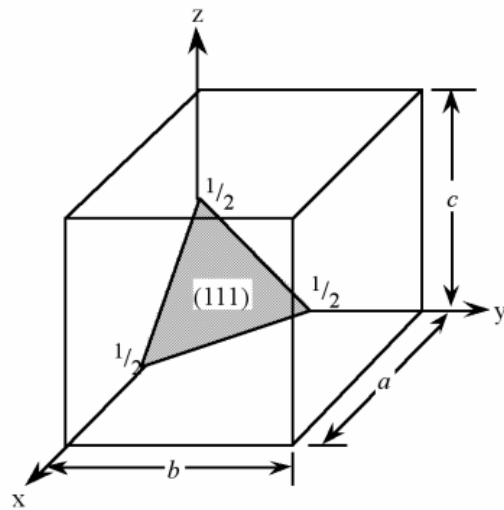


Diffracting planes in a crystal



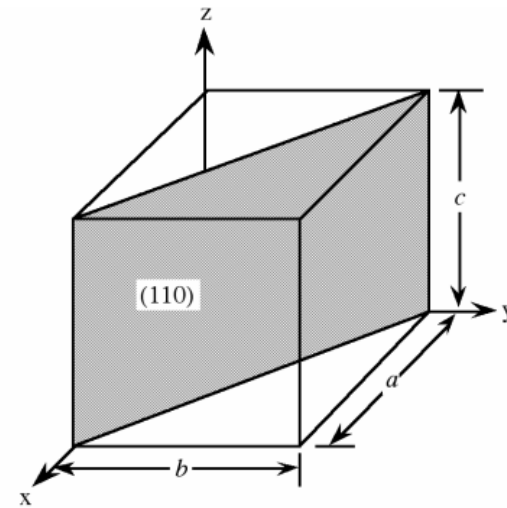
$$h \quad k \quad l$$

$$\frac{1}{1}, \frac{1}{\infty}, \frac{1}{\infty} = (100)$$



$$h \quad k \quad l$$

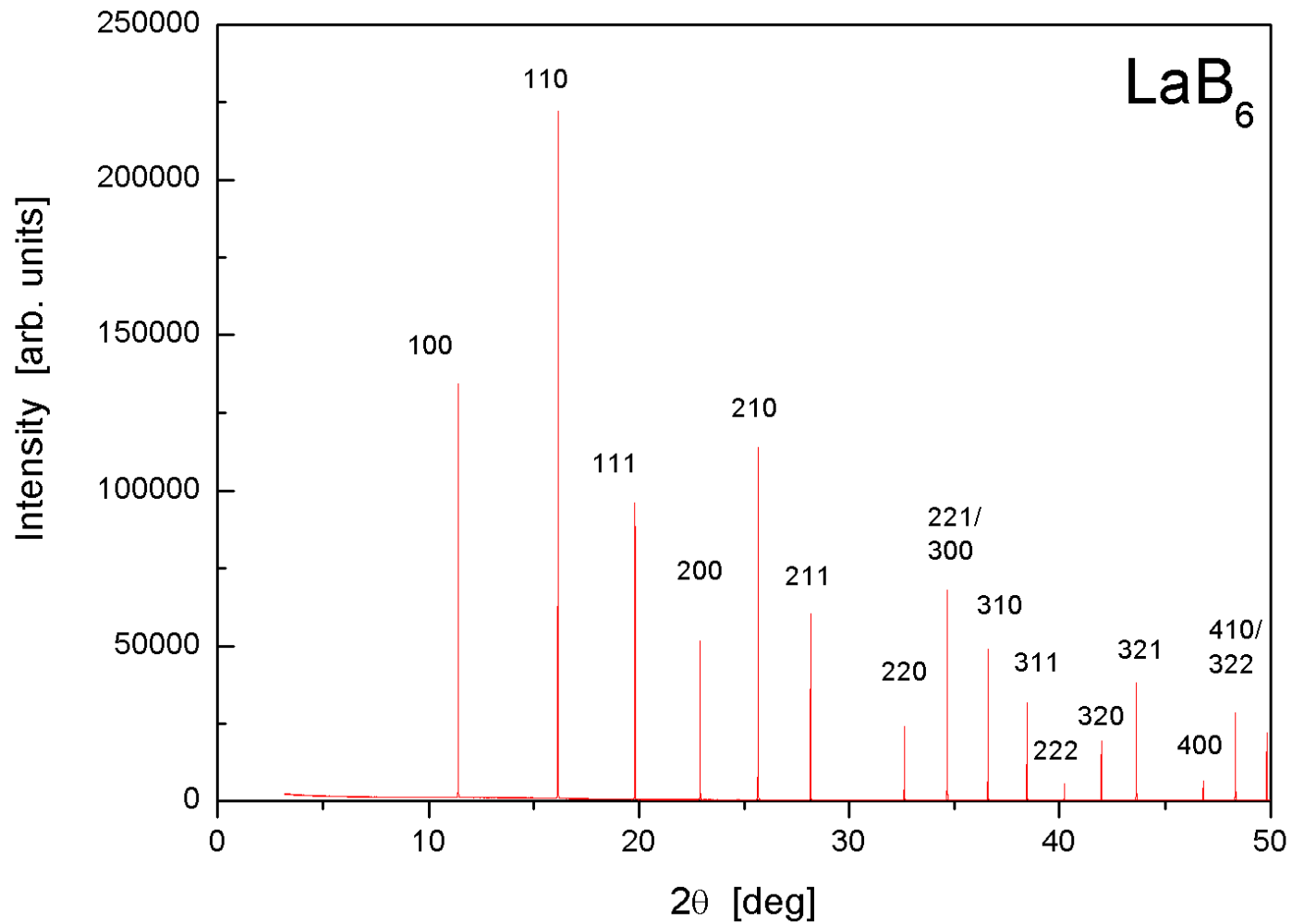
$$\frac{1}{1/2}, \frac{1}{1/2}, \frac{1}{1/2} = (222) = (111)$$



$$h \quad k \quad l$$

$$\frac{1}{1}, \frac{1}{1}, \frac{1}{\infty} = (110)$$

Peak positions



Peak position:

$$d^2 = \frac{h^2+k^2+l^2}{a^2}$$

$$d = n\lambda/2\sin\theta$$

Diffraction intensity

$$I_{(hkl)\alpha} = \frac{I_0 \lambda^3}{64\pi r} \left(\frac{e^2}{m_e c^2} \right)^2 \frac{M_{(hkl)}}{V_\alpha^2} |F_{(hkl)\alpha}|^2 \left(\frac{1 + \cos^2(2\theta) \cos^2(2\theta_m)}{\sin^2 \theta \cos \theta} \right)_{hkl} \frac{v_\alpha}{\mu_s}$$

Geometrical

Electron scattering power

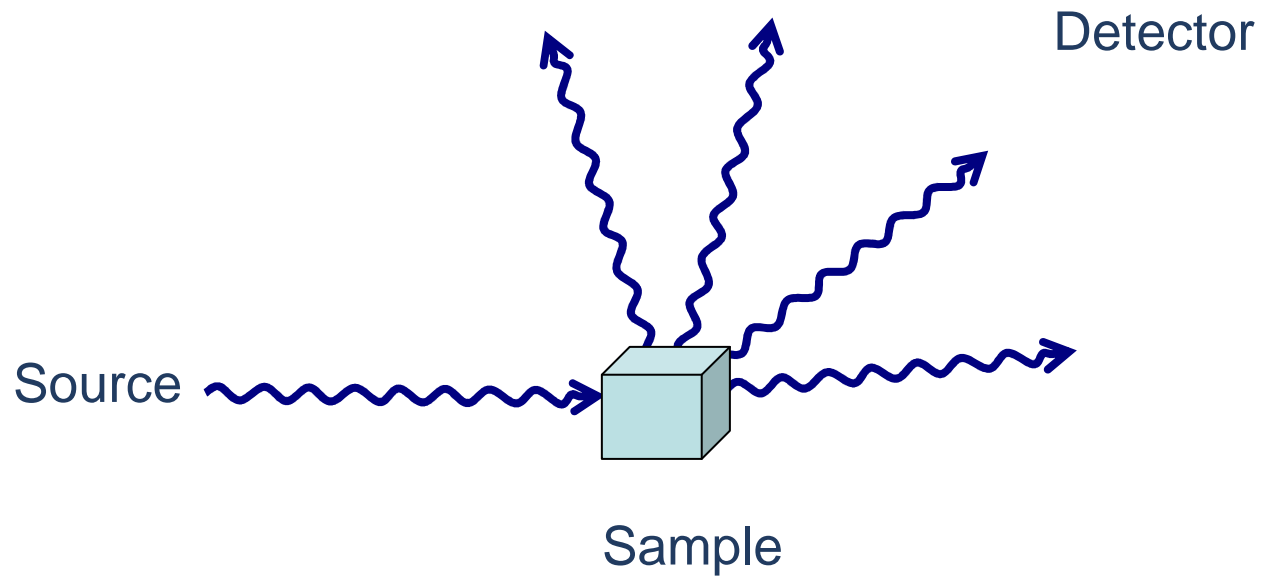
Symmetry

Structure factor

X-ray polarisation factor

Sample absorbance

X-ray diffraction experiment setup



Different sources



IRL laboratory instrument



Australian Synchrotron

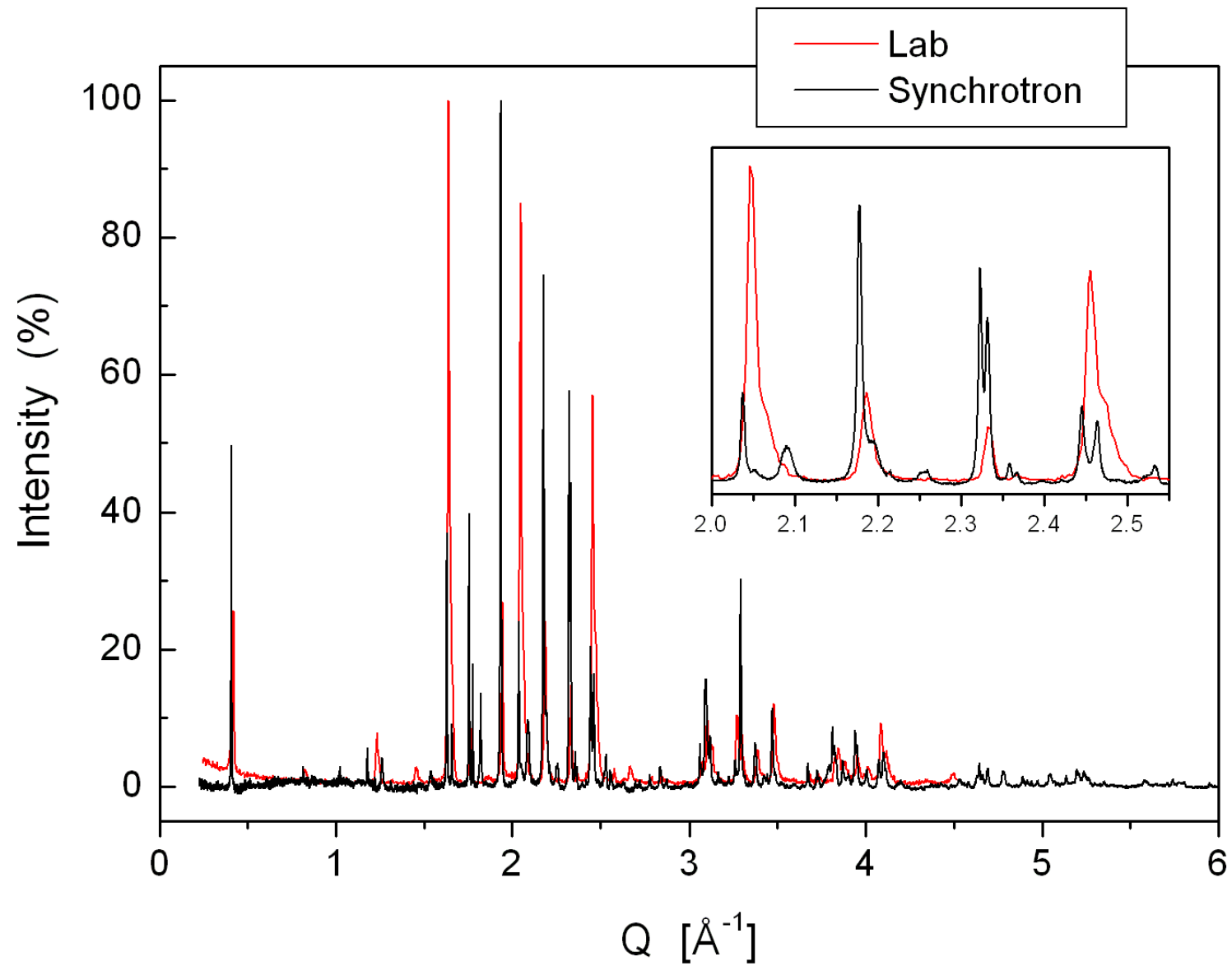
Photon flux ratio 1:1,000,000

Advantages of using a synchrotron source for X-ray diffraction

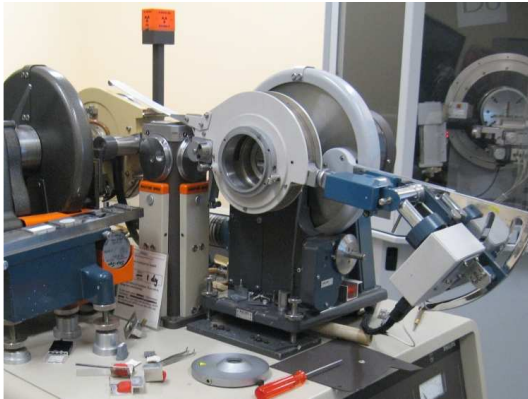
- Increased flux
 - Smaller samples
 - Weakly scattering samples
 - Poorly crystalline, low Z
 - Faster measurement
 - Real-time experiments
 - High throughput
- Lower divergence
 - Better resolution

Question: Which would be suitable for X-ray diffraction: bend magnet, wiggler, or undulator?

Instrument resolution

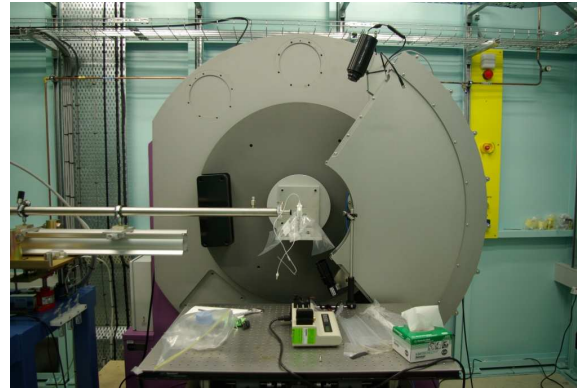


Different detectors



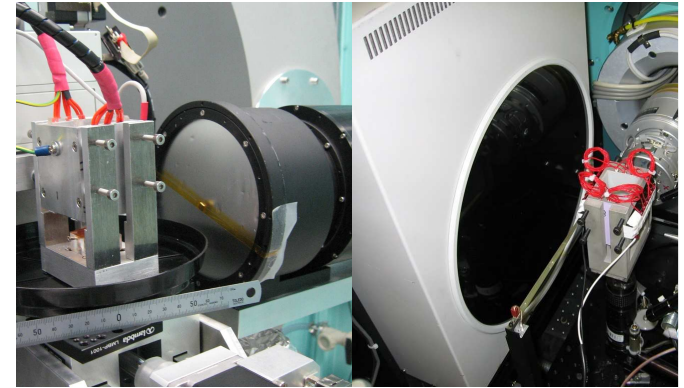
Point detector

- high resolution
- post-sample collimation: grazing incidence
- \$
- slow



Strip detector

- collect entire pattern at once
- fast readout
- \$\$
- no post-sample collimation



Area detector

- collect entire pattern at once
- fast, direct observation of texture
- \$\$\$
- limited resolution
- no post-sample collimation

Different types of samples: single crystal

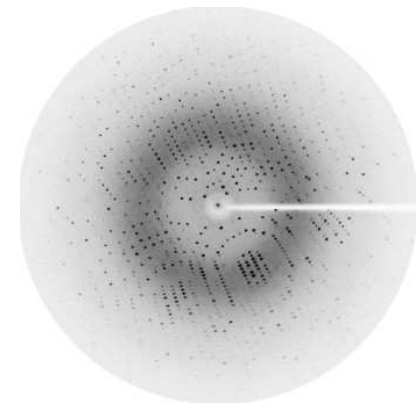
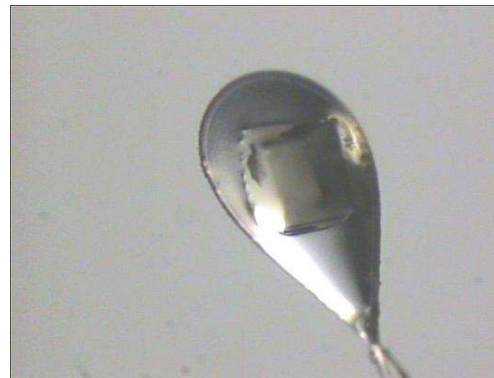


- ●
- ●
- ●

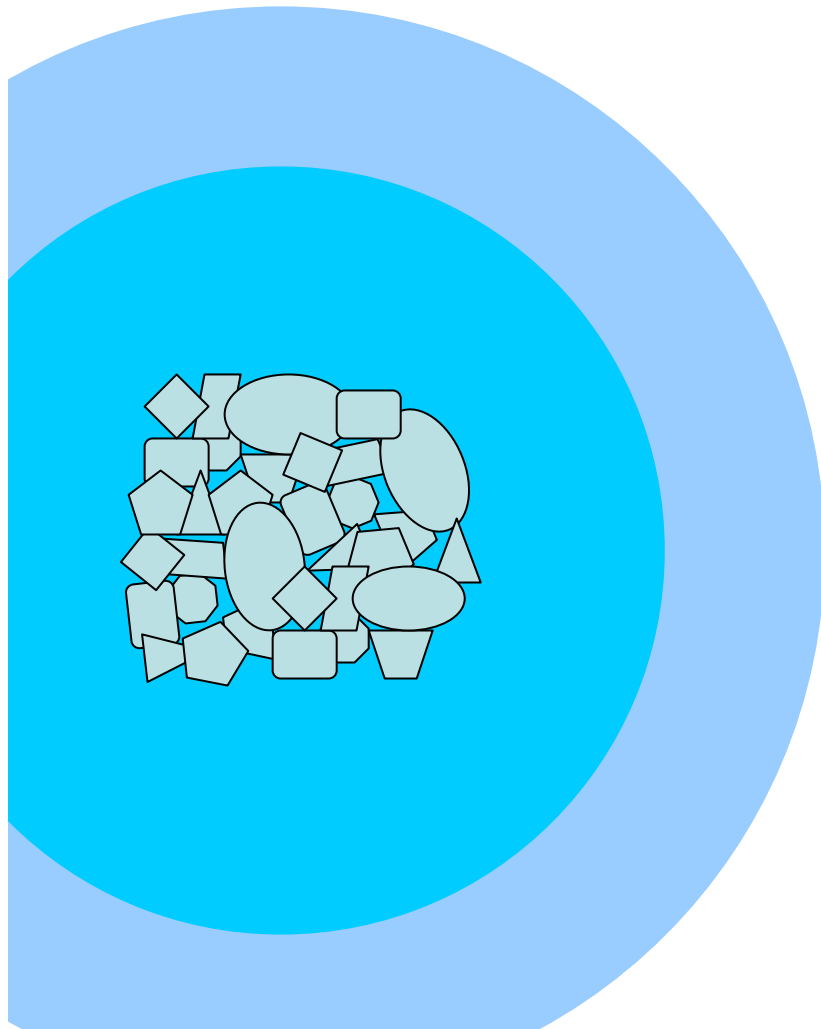
One large crystallite

Diffraction condition is met at points in 3D reciprocal space

Measure using 2D detector, rotate sample



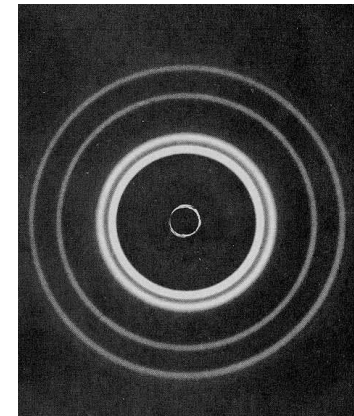
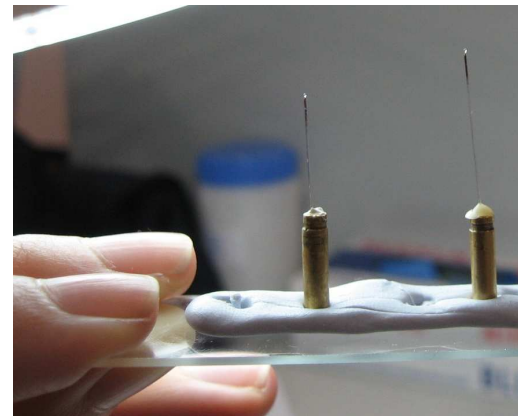
Different types of samples: powder



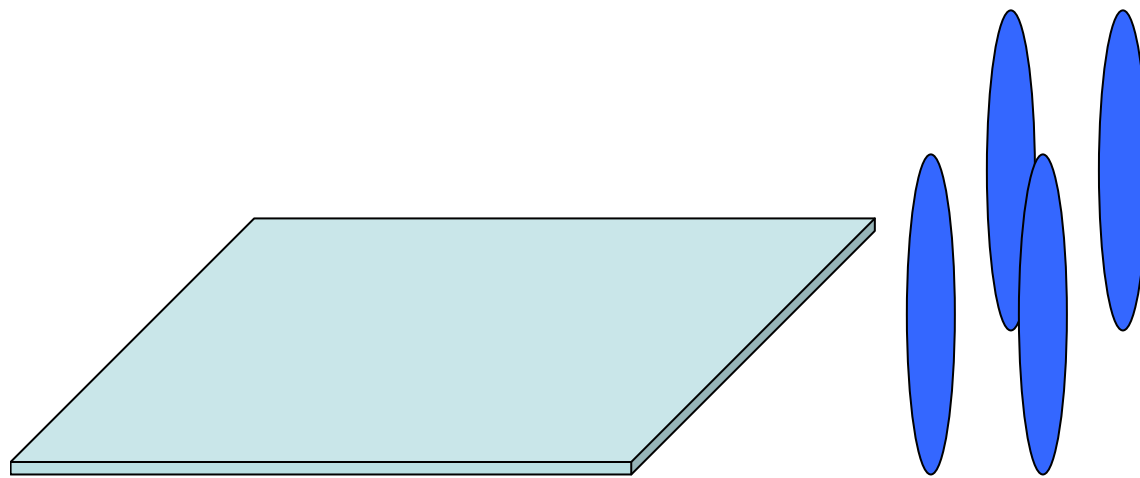
Many small crystallites,
randomly oriented

Diffraction condition is met
at spheres in 3D reciprocal
space - isotropic

Measure in one direction
(point or strip detector ok)

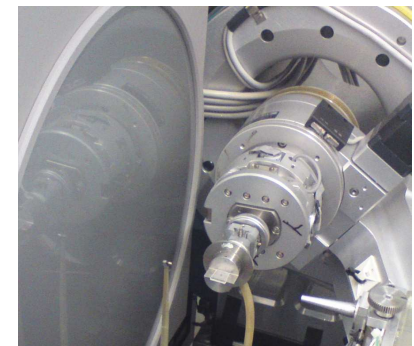
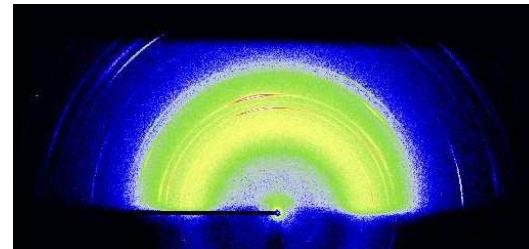
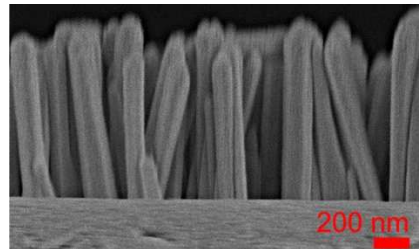


Different types of samples: thin film



May be oriented out of plane (isotropic in plane) or epitaxial

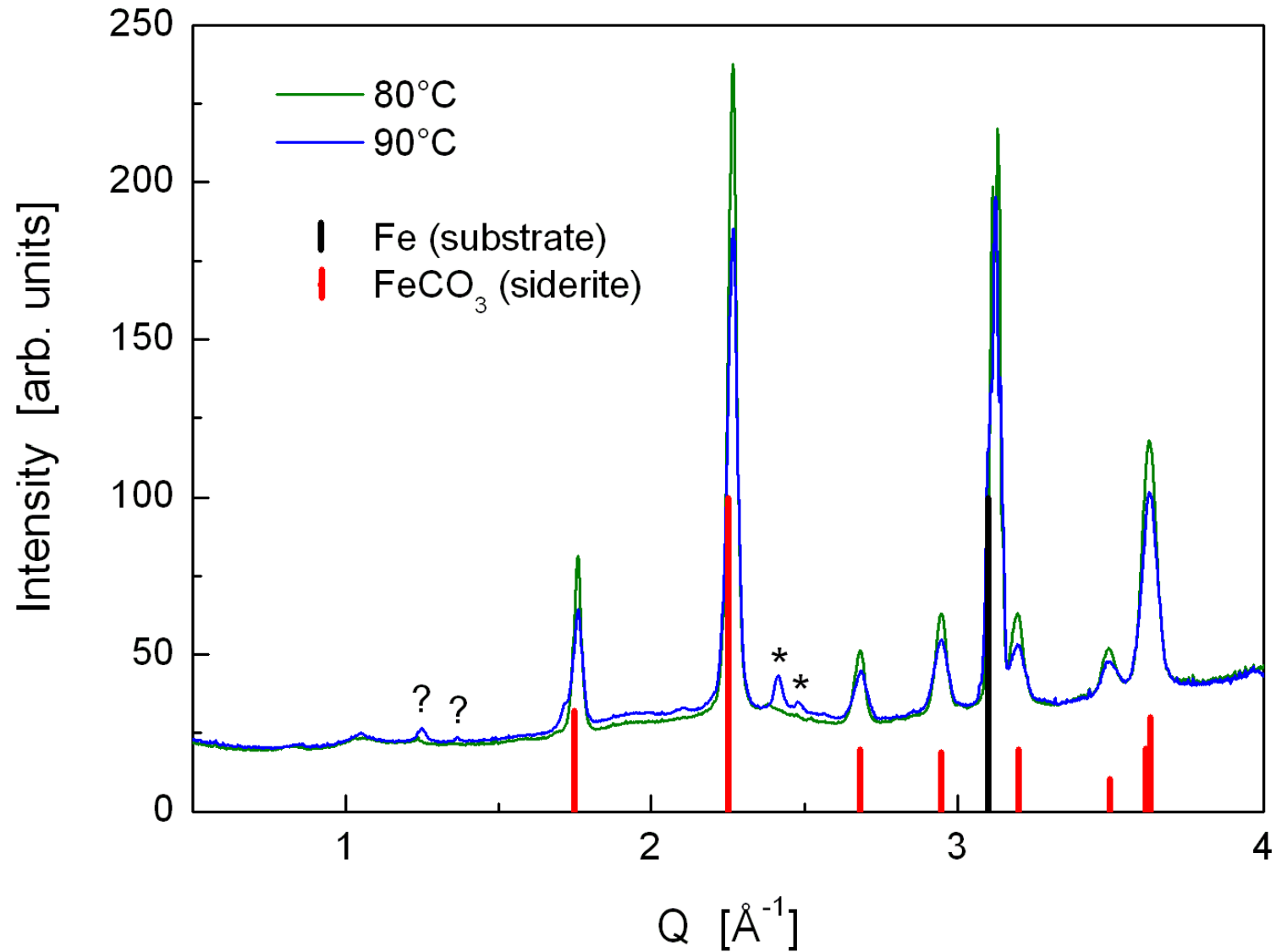
Normally use grazing incidence to reduce substrate diffraction

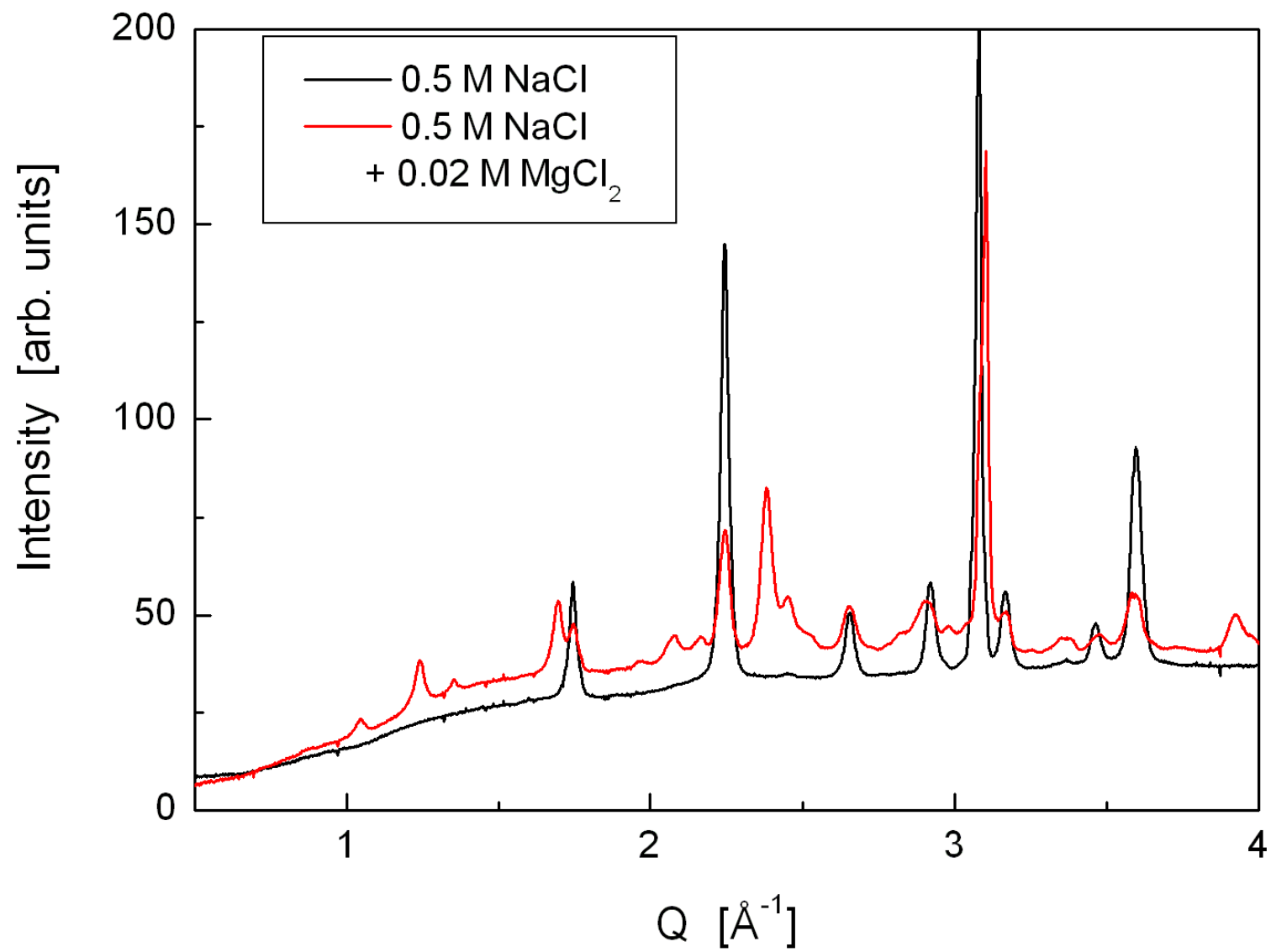


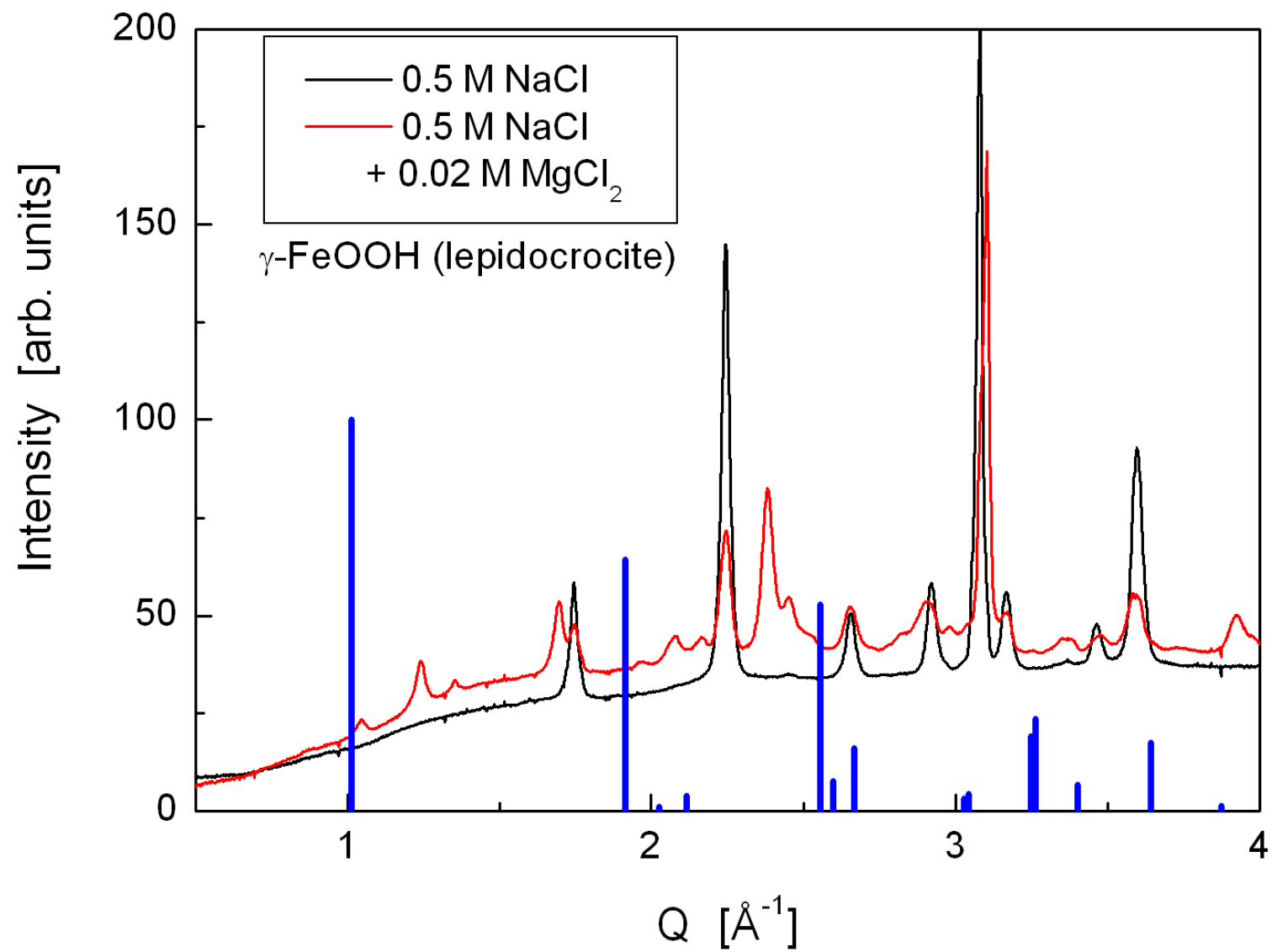
What can we find out?

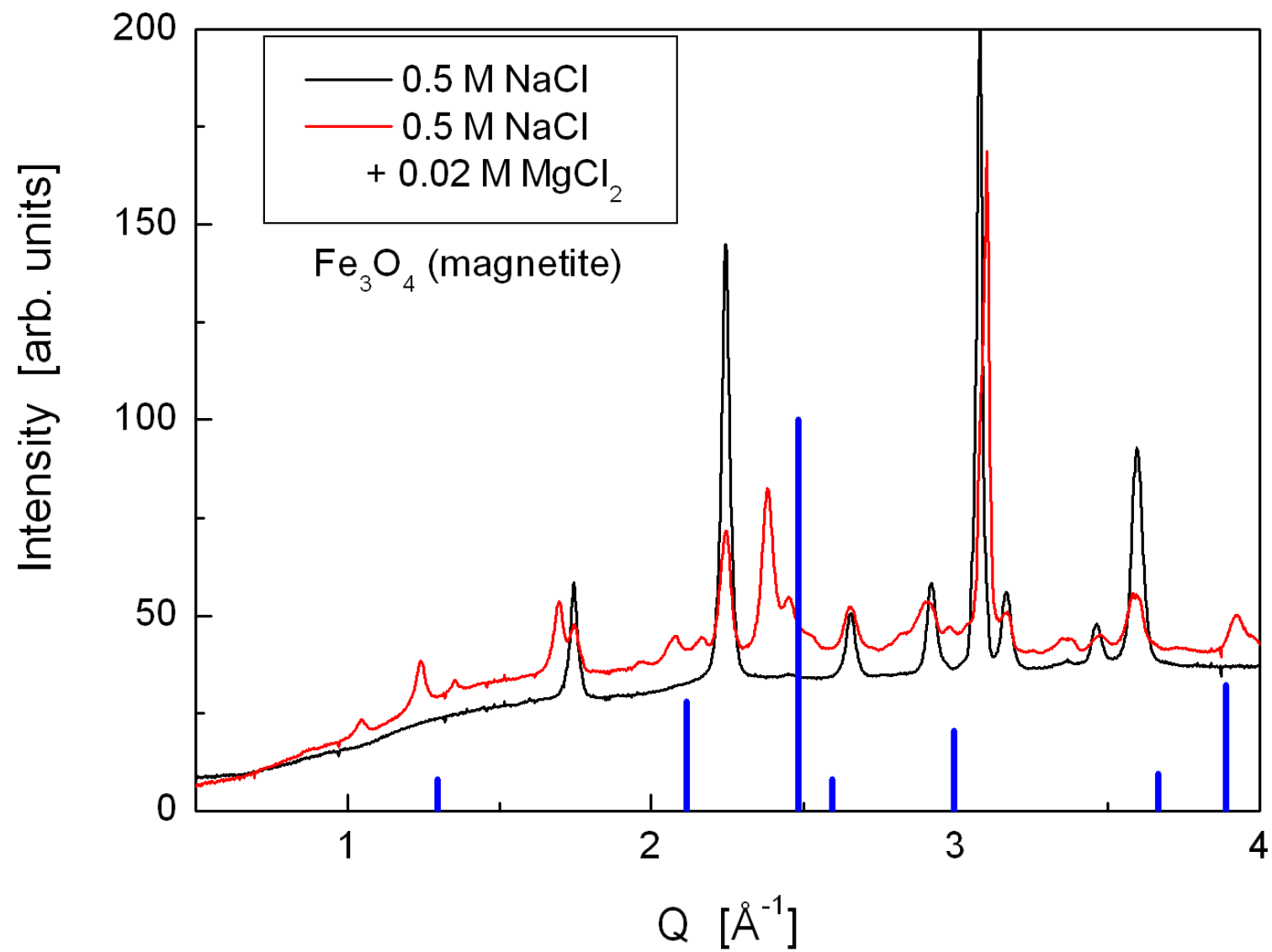
- Phases
 - Composition, ratio
- Peak positions: lattice parameters
- Peak width: crystallite size, strain
- Full pattern analysis
 - Accurate lattice parameters, atomic positions
- Texture (preferential orientation)
 - Direction and degree of texture

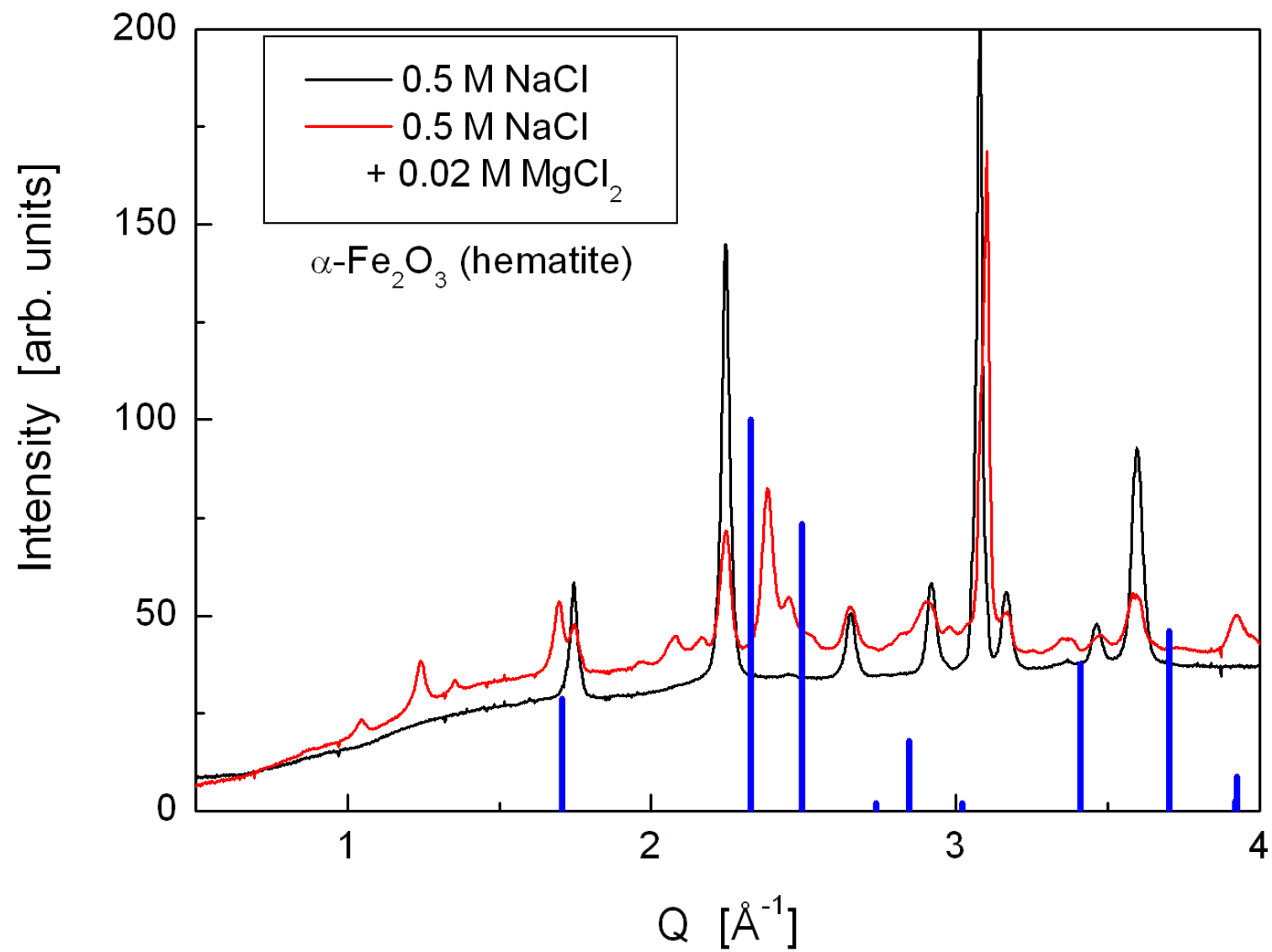
1. Phase analysis

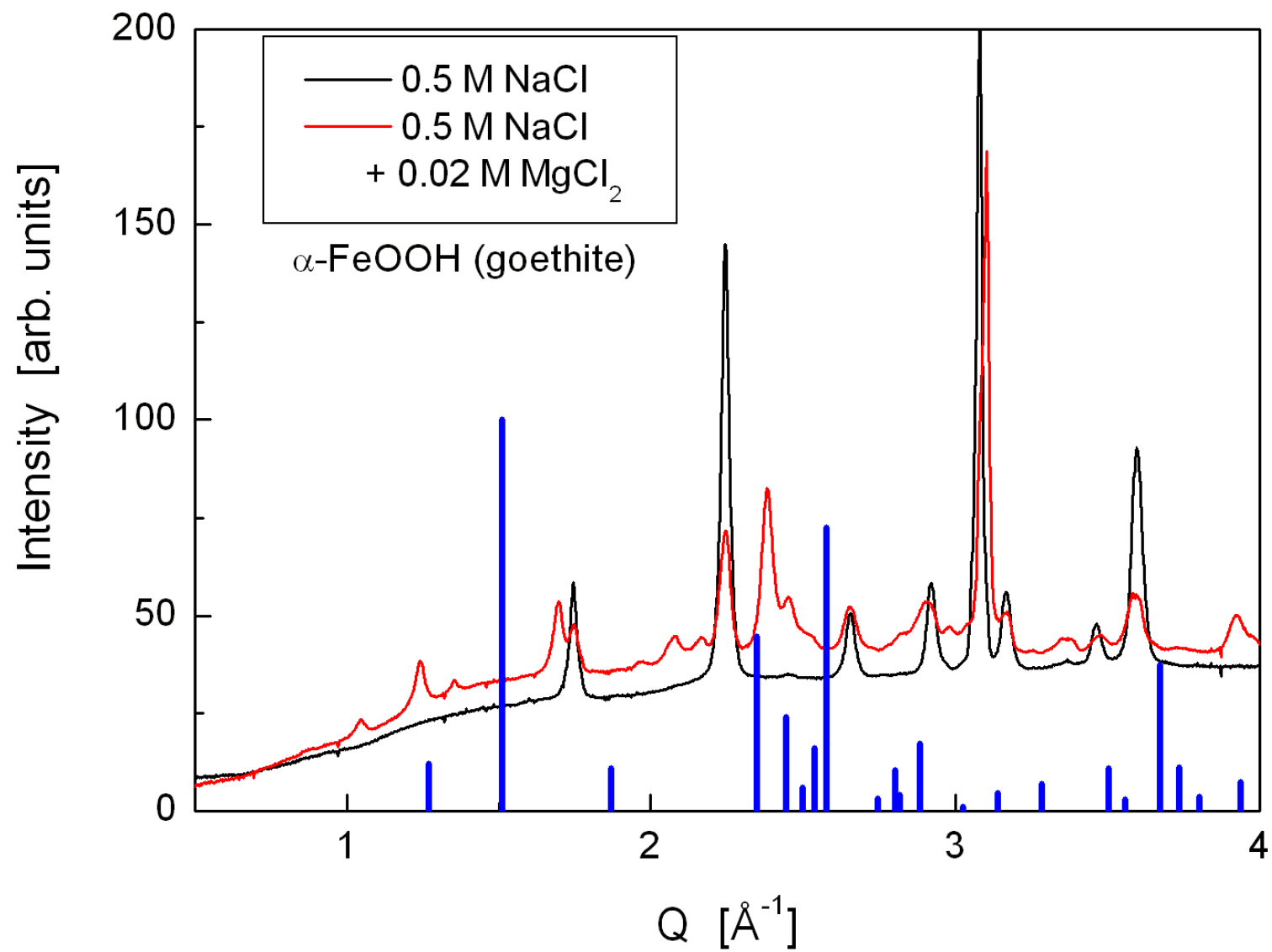


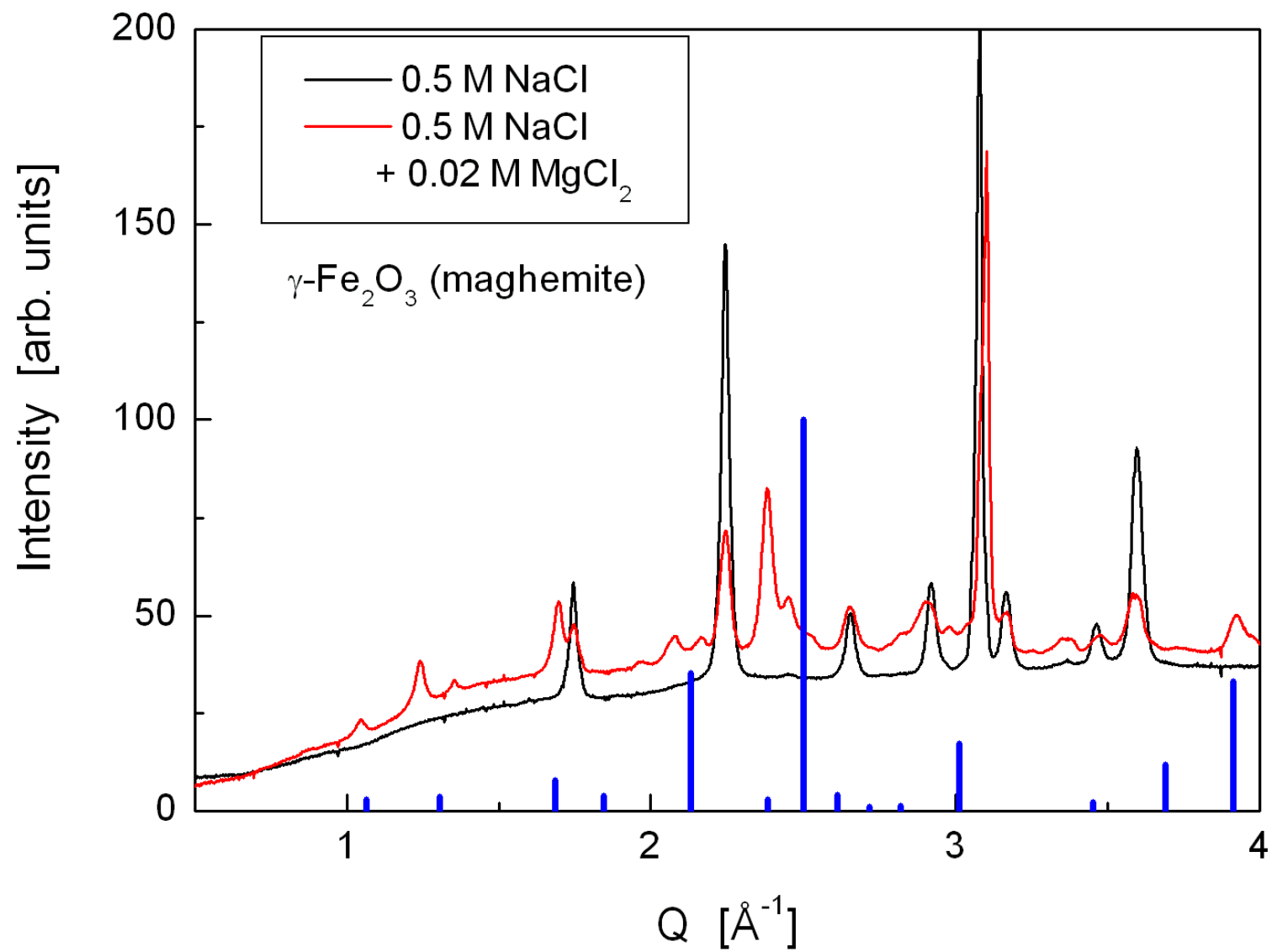


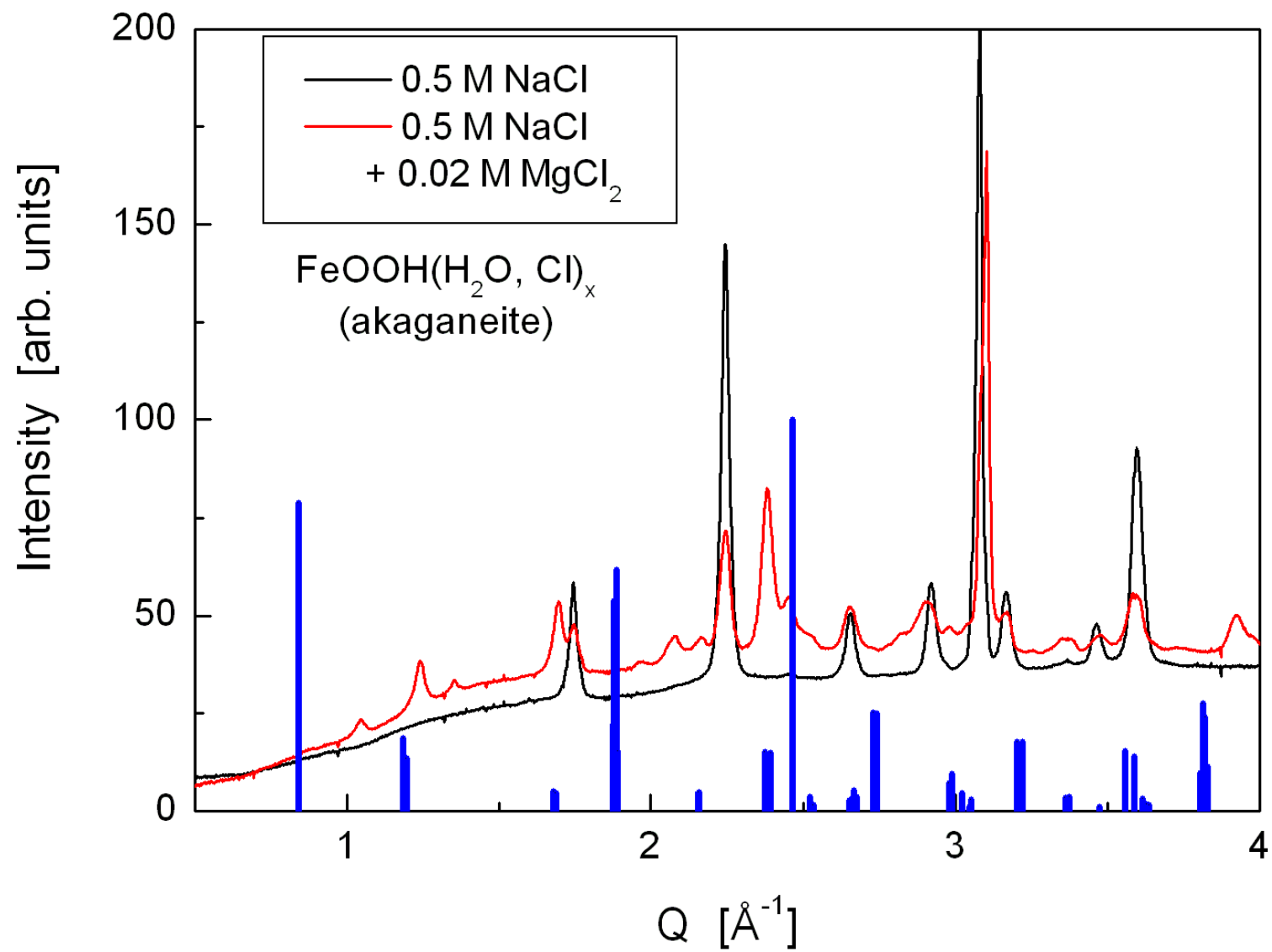


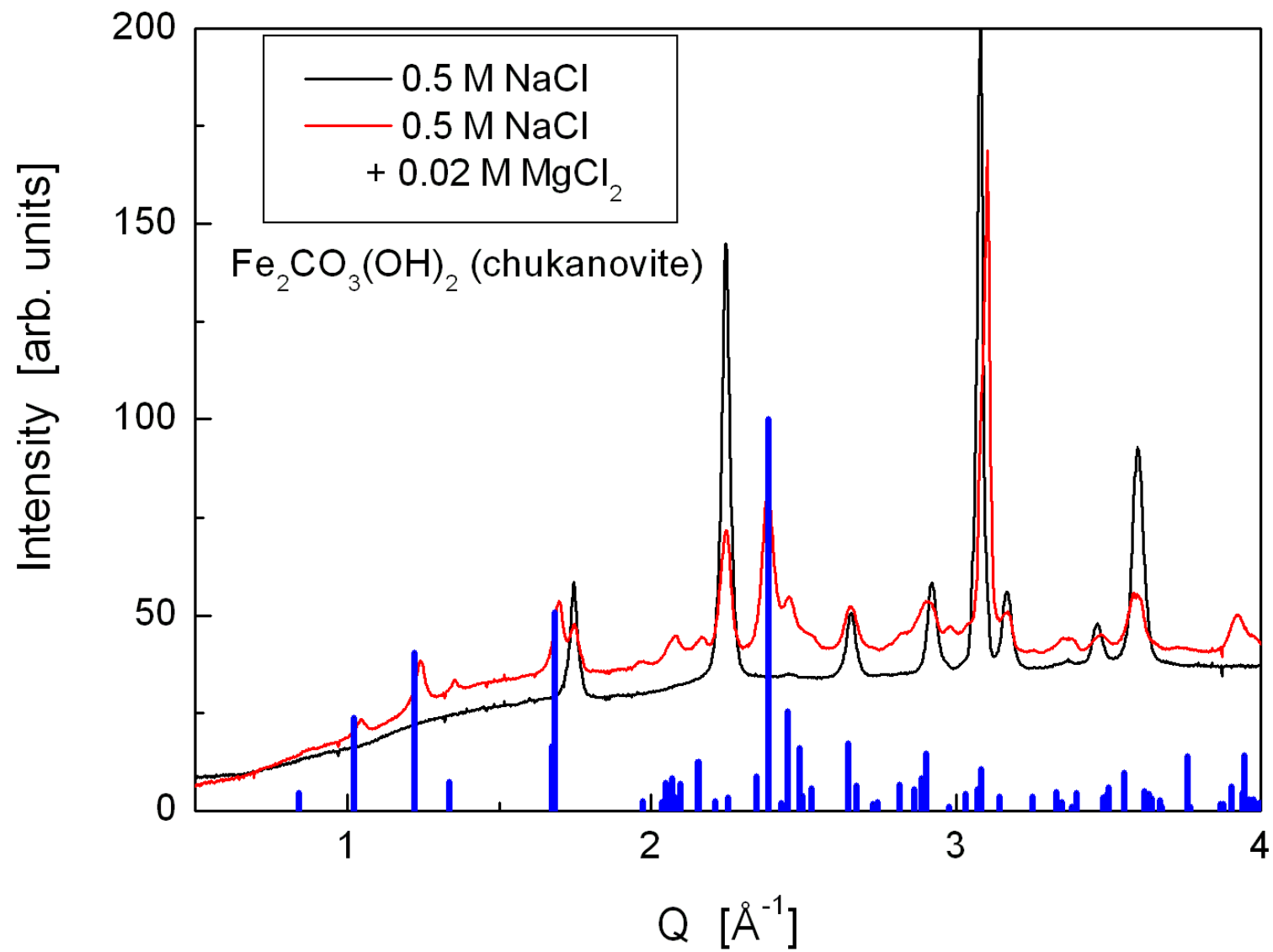




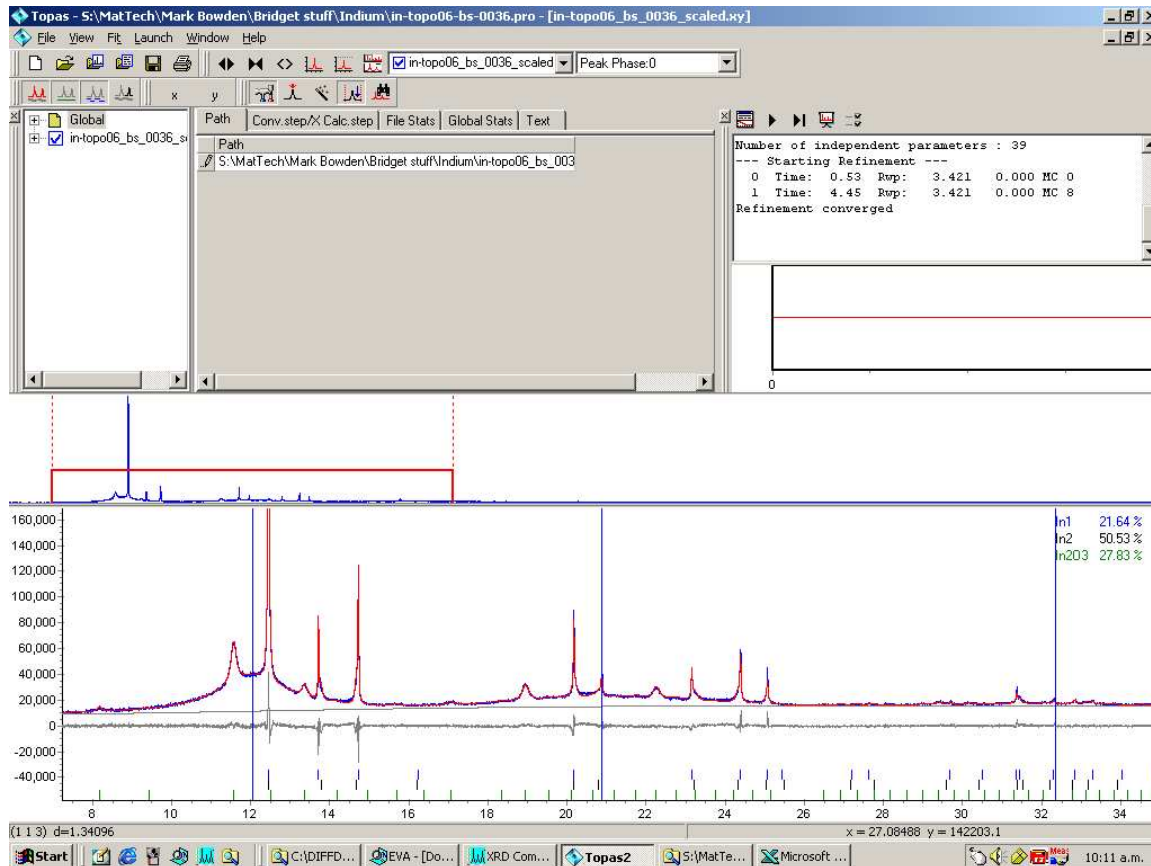








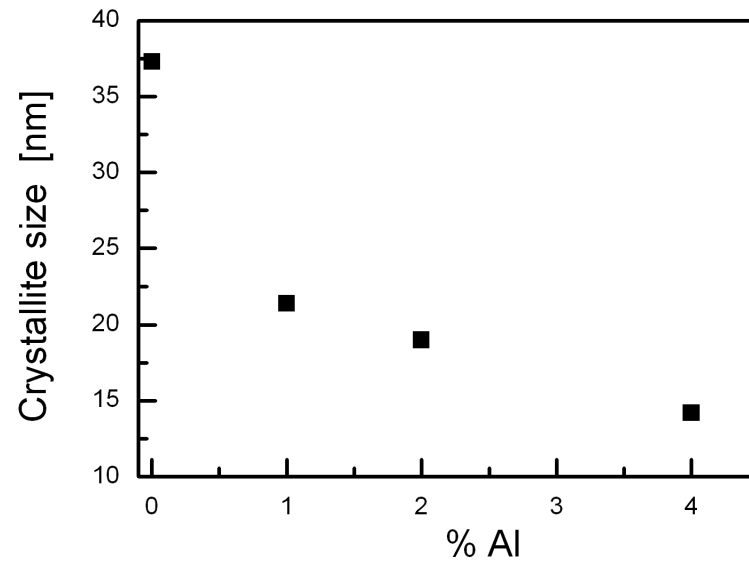
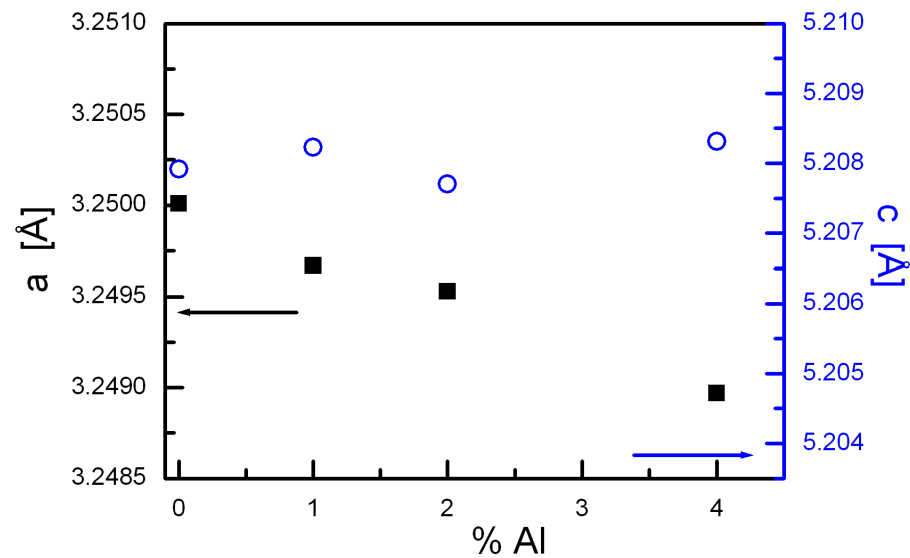
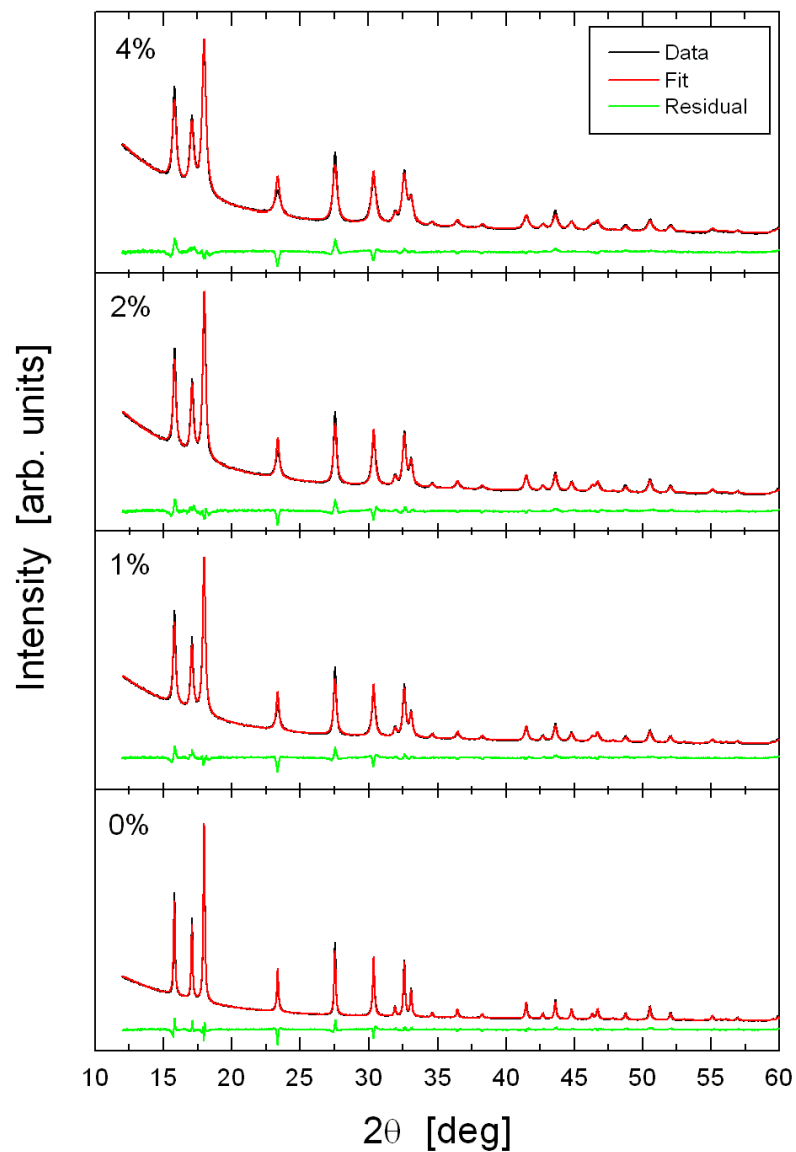
2. Full pattern analysis



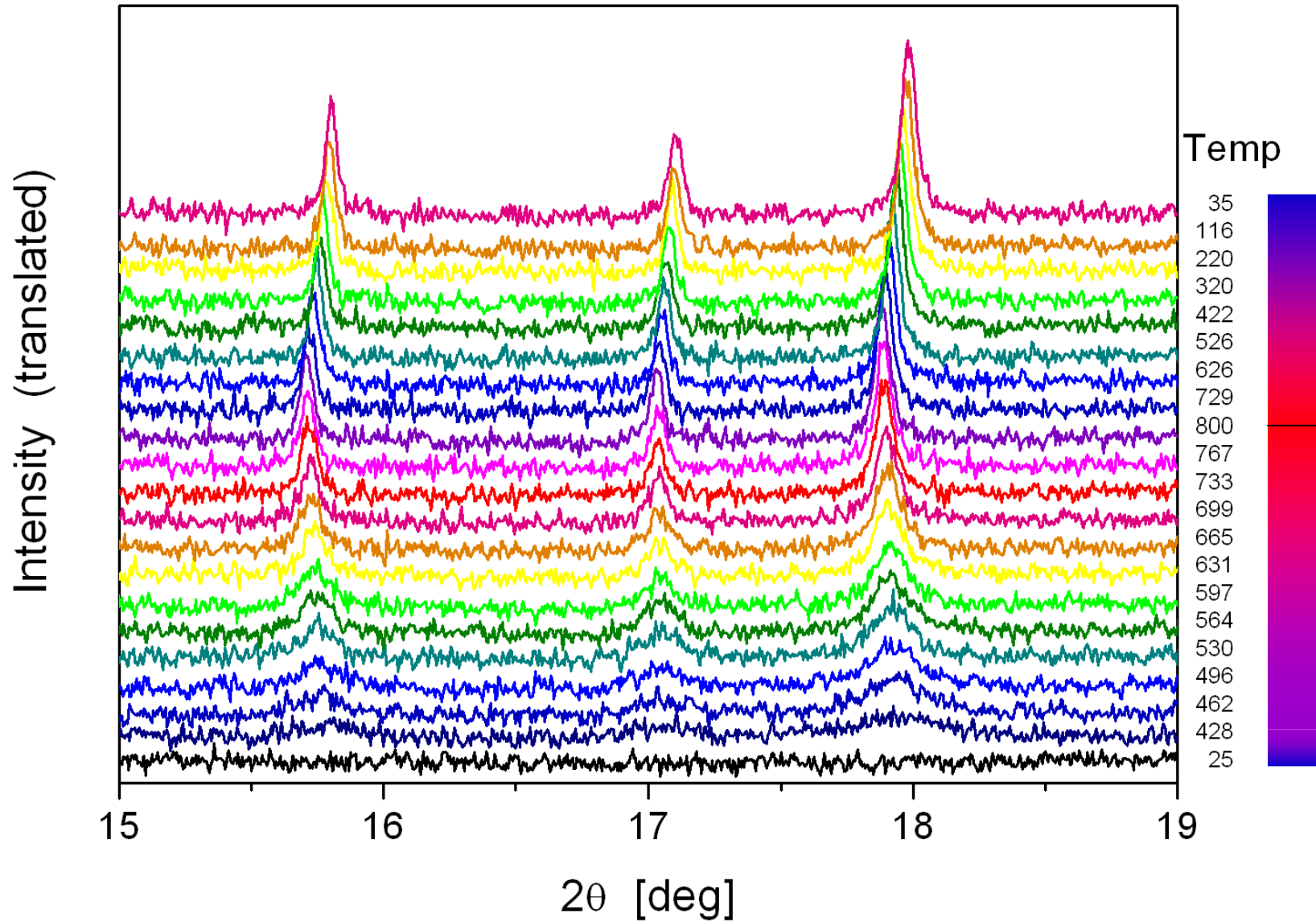
Using standard packages
(e.g. GSAS, TOPAS)

Obtain accurate lattice
parameters, crystallite size,
strain, atomic positions
(Rietveld refinement)

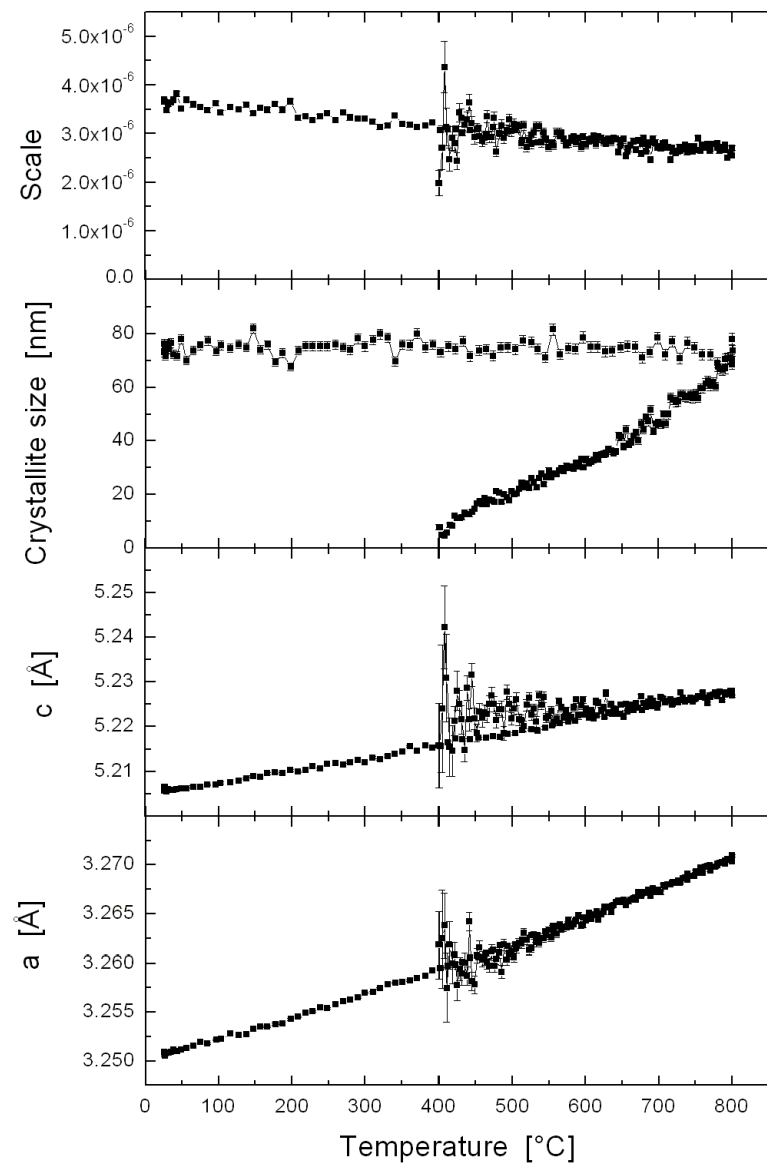
Case study: Al-doped ZnO



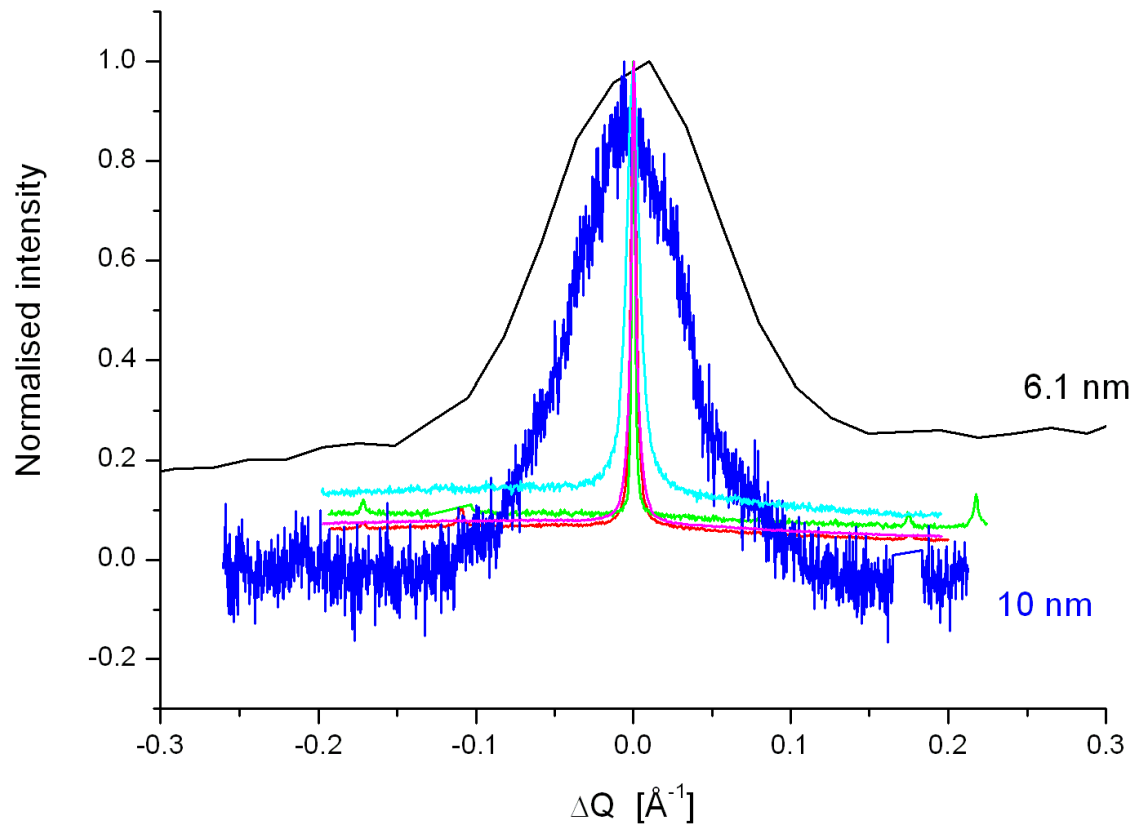
Temperature/time series



Temperature/time series results



3. Peak width: crystallite size, deformations

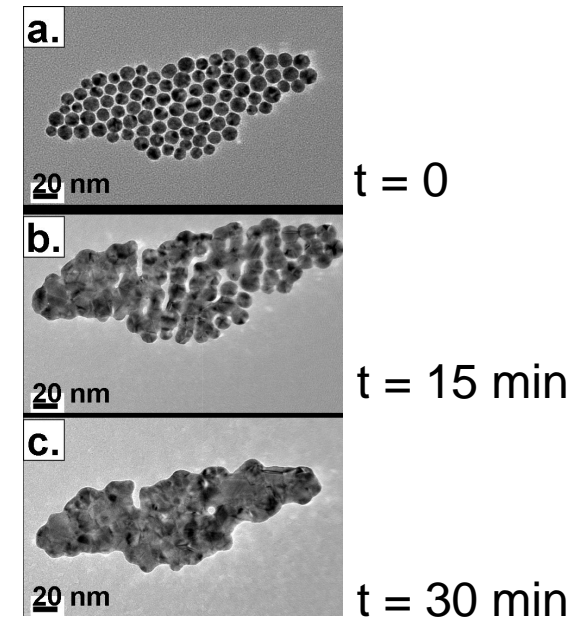
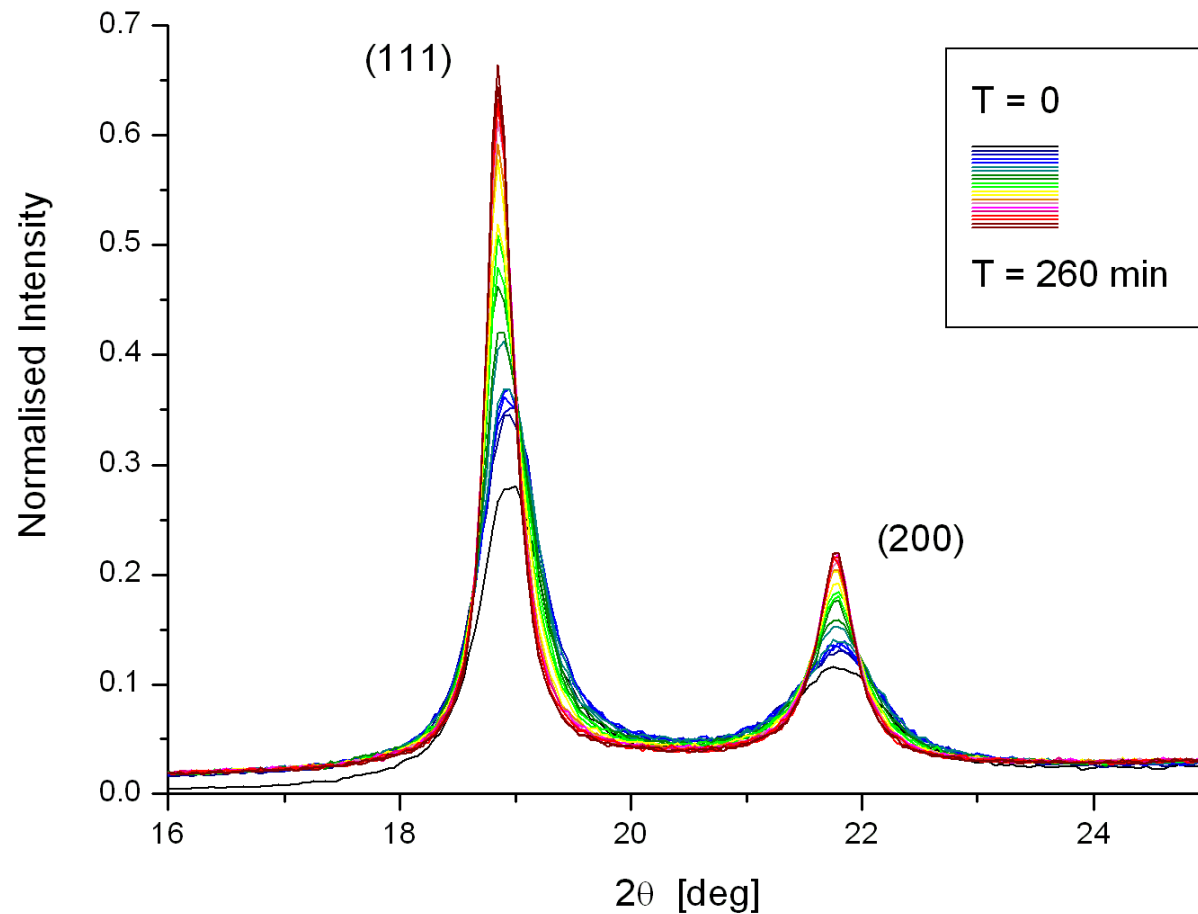


Peaks are broadened by crystal defects: finite size, deformations

Scherrer equation:

$$\text{Crystallite size} = \frac{0.9\lambda}{FW\cos\theta}$$

Case study: Au nanoparticle coalescence



Observe:

- Peaks become sharper
 - grain growth
- Peaks move apart
 - release of strain (fault density decreases)
- Peak area constant
 - no melting, transformation

Analysis

For *fcc* metals,
$$\Delta(2\theta_{200} - 2\theta_{111})^\circ = \frac{-90\sqrt{3}\alpha}{\pi^2} \left(\frac{\tan \theta_{200}}{2} + \frac{\tan \theta_{111}}{4} \right)$$

For the (111) peak,
$$\frac{1}{D_{eff(111)}} = \frac{1}{D} + \frac{3}{16} \frac{(1.5\alpha + \beta)}{a}$$

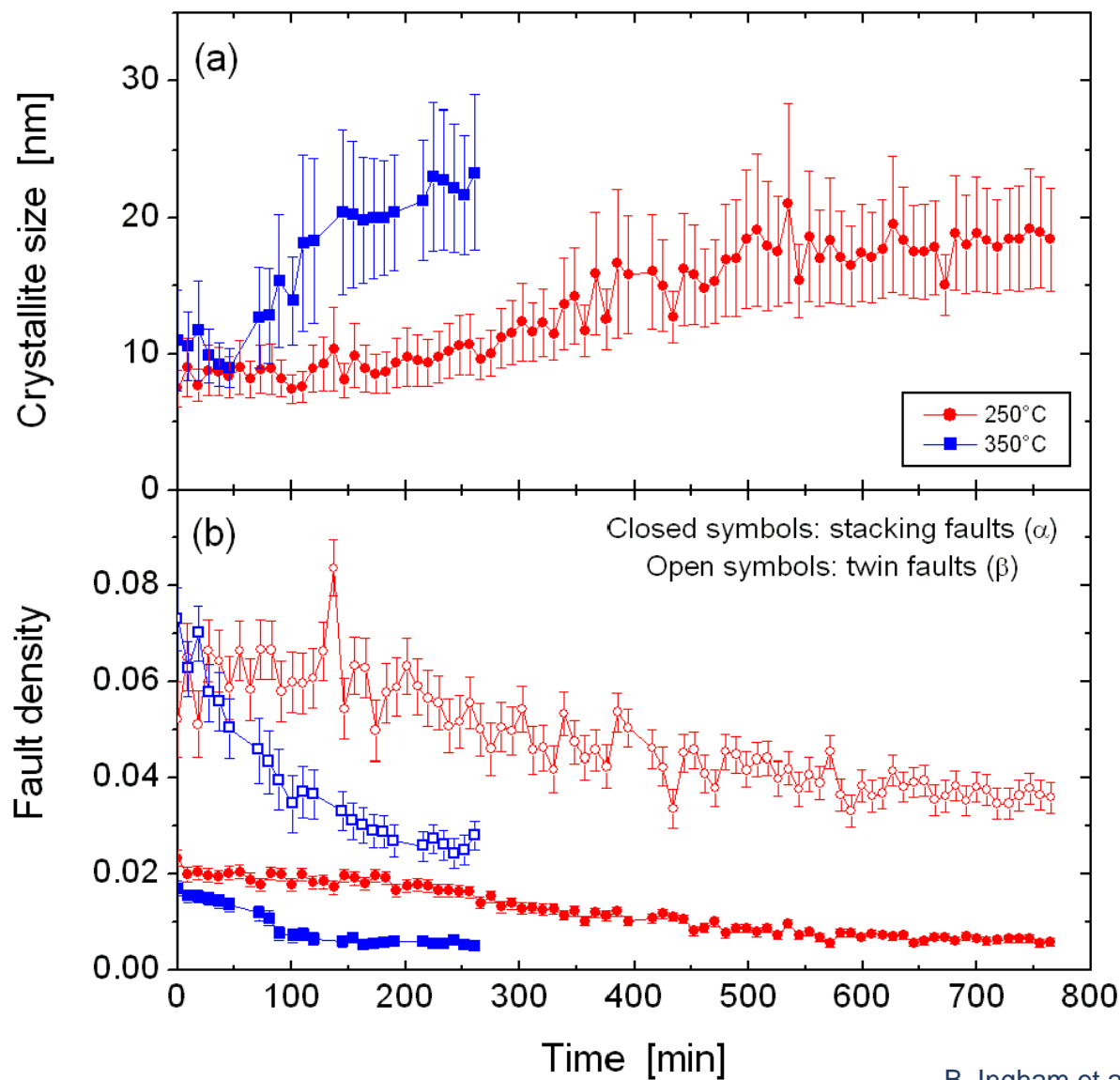
For the (200) peak,
$$\frac{1}{D_{eff(200)}} = \frac{1}{D} + \frac{(1.5\alpha + \beta)}{a}$$

B. E. Warren,
X-ray Diffraction,
Dover Publishers
(1969)

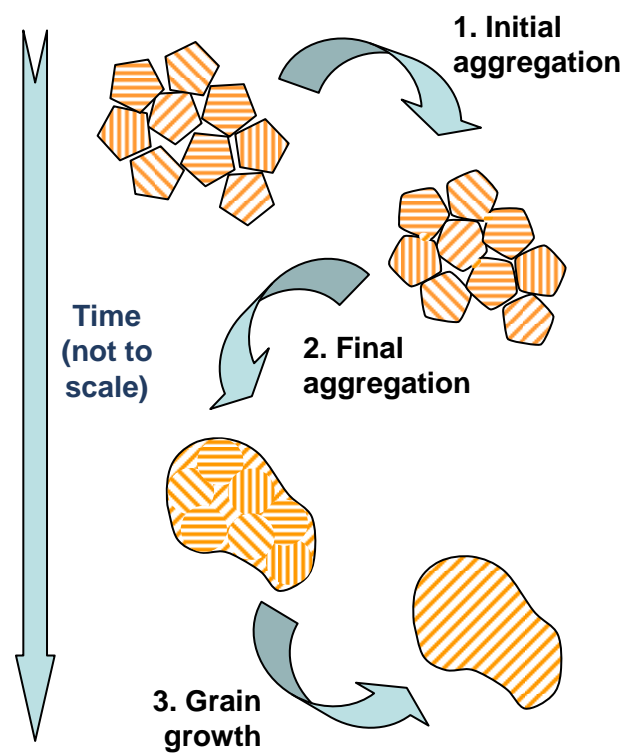
Therefore the deformation fault density α , the twin fault density β , and the true crystallite size D can be obtained by comparing the peak separation and peak widths together.



Results

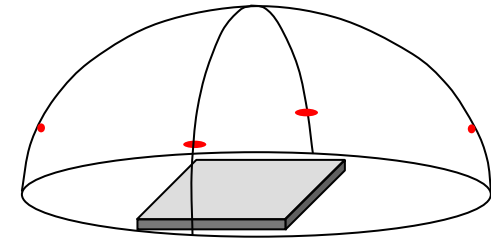
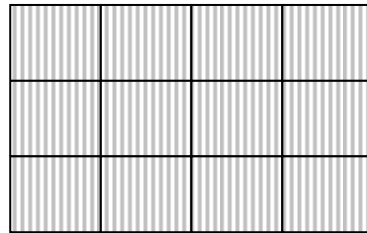
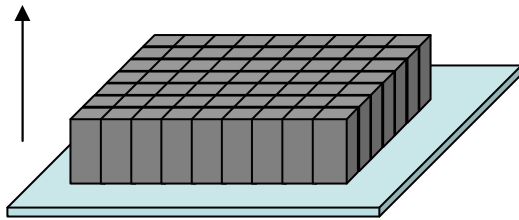


Combine with SAXS to compare the difference between particle size and crystallite size vs. time

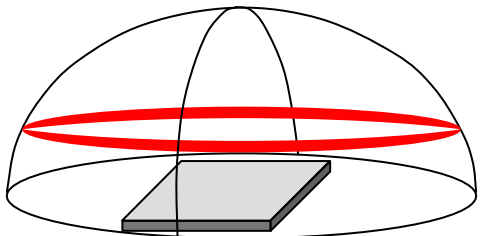
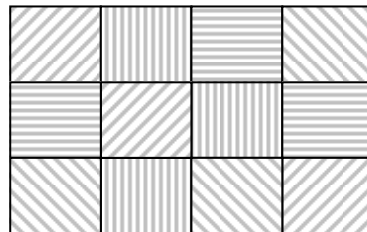
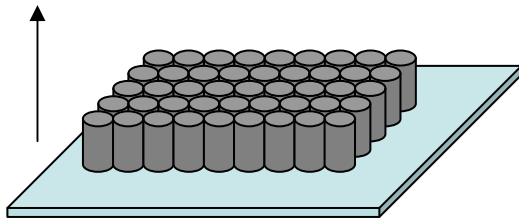


4. Texture (preferential orientation)

Epitaxial



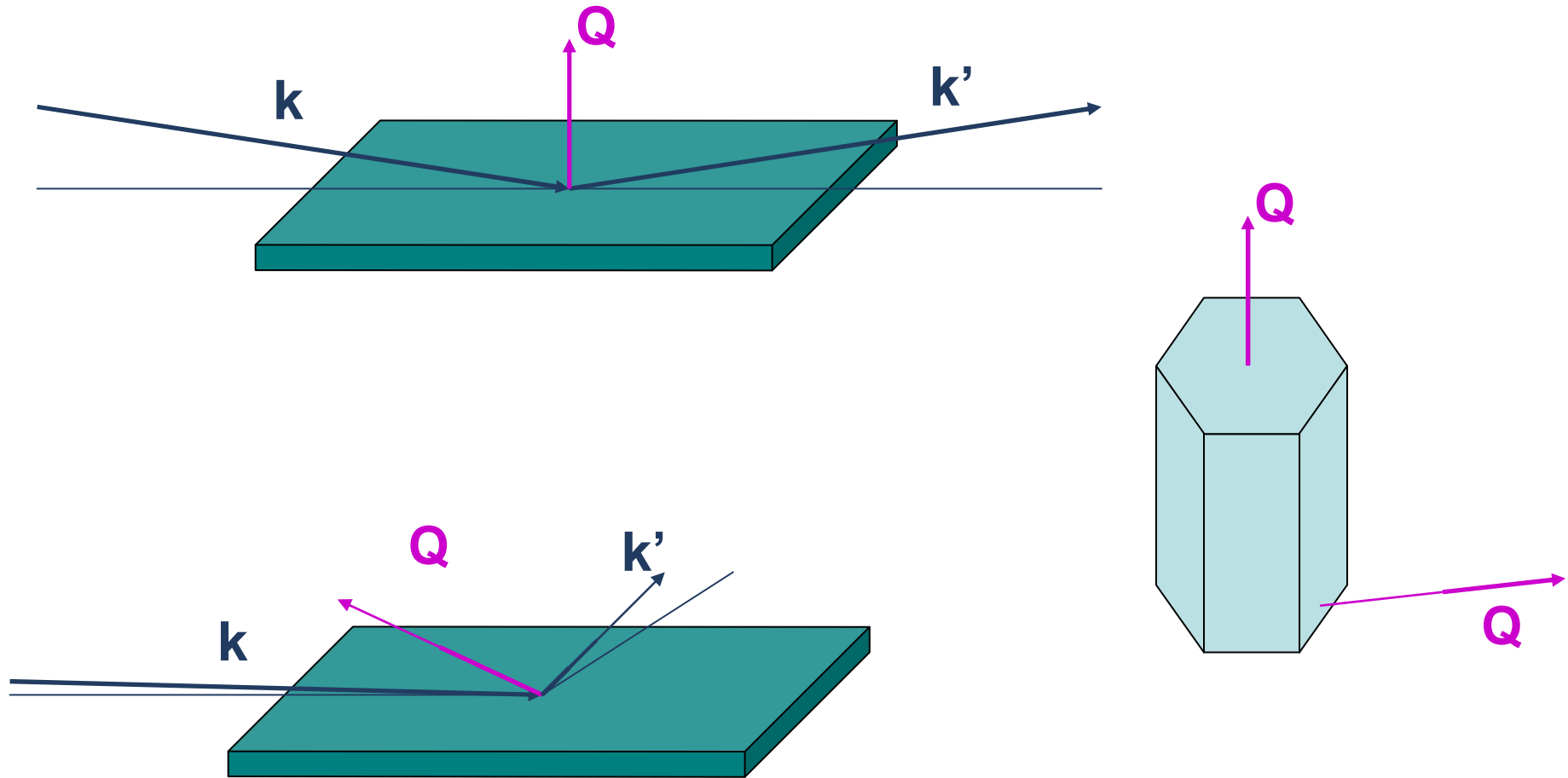
Fibre texture



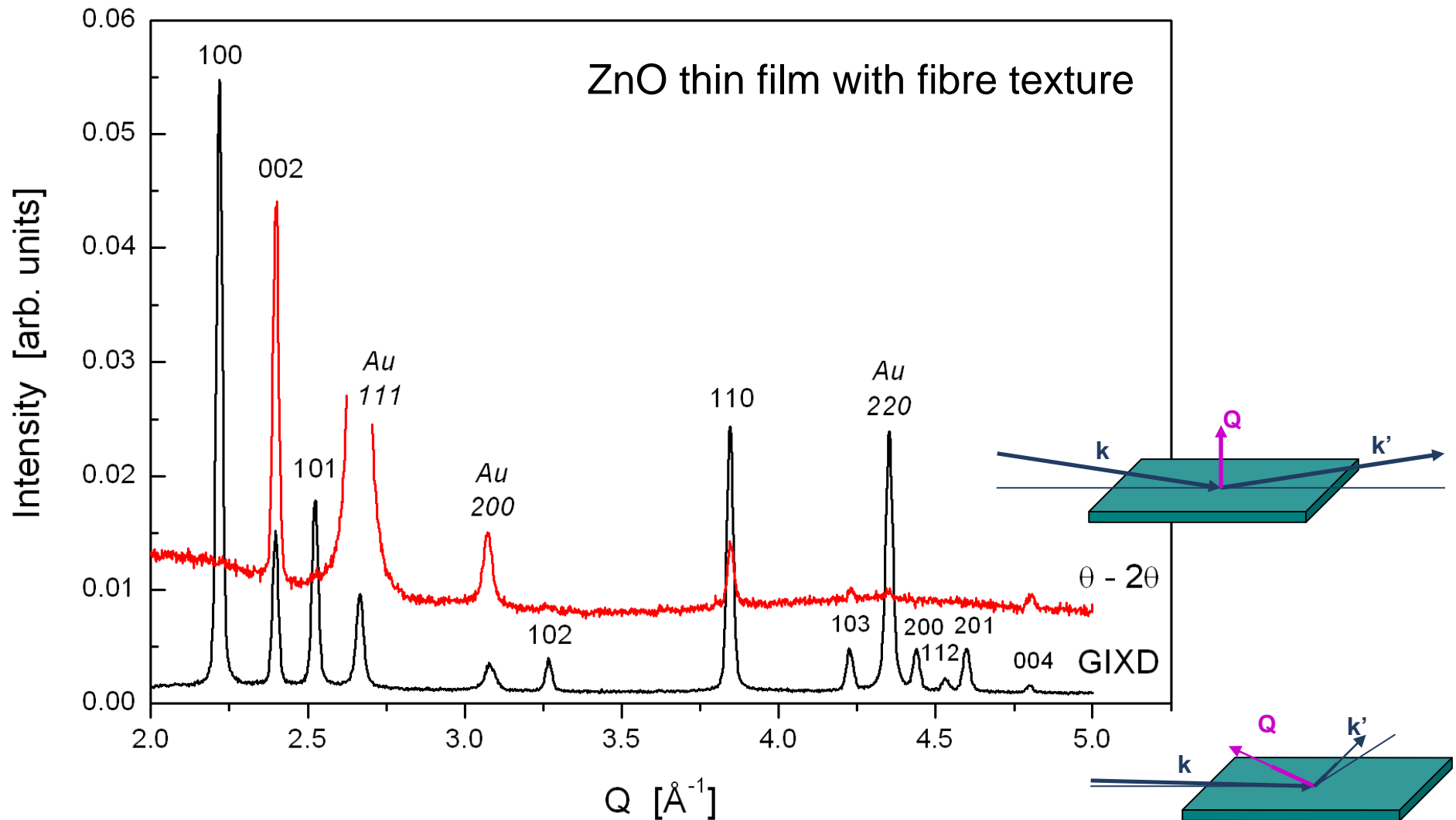
Top view

Diffraction

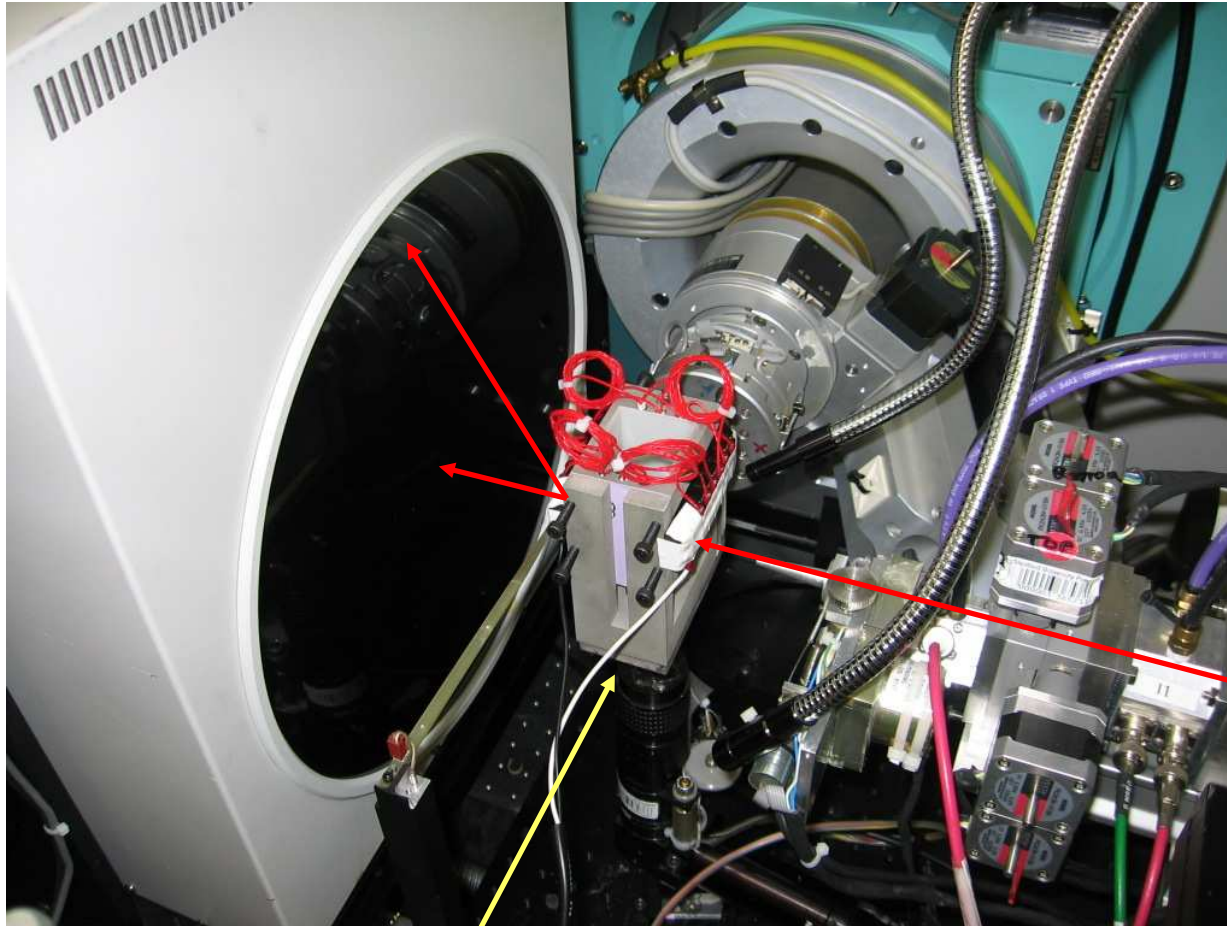
Different XRD geometries



Different XRD geometries

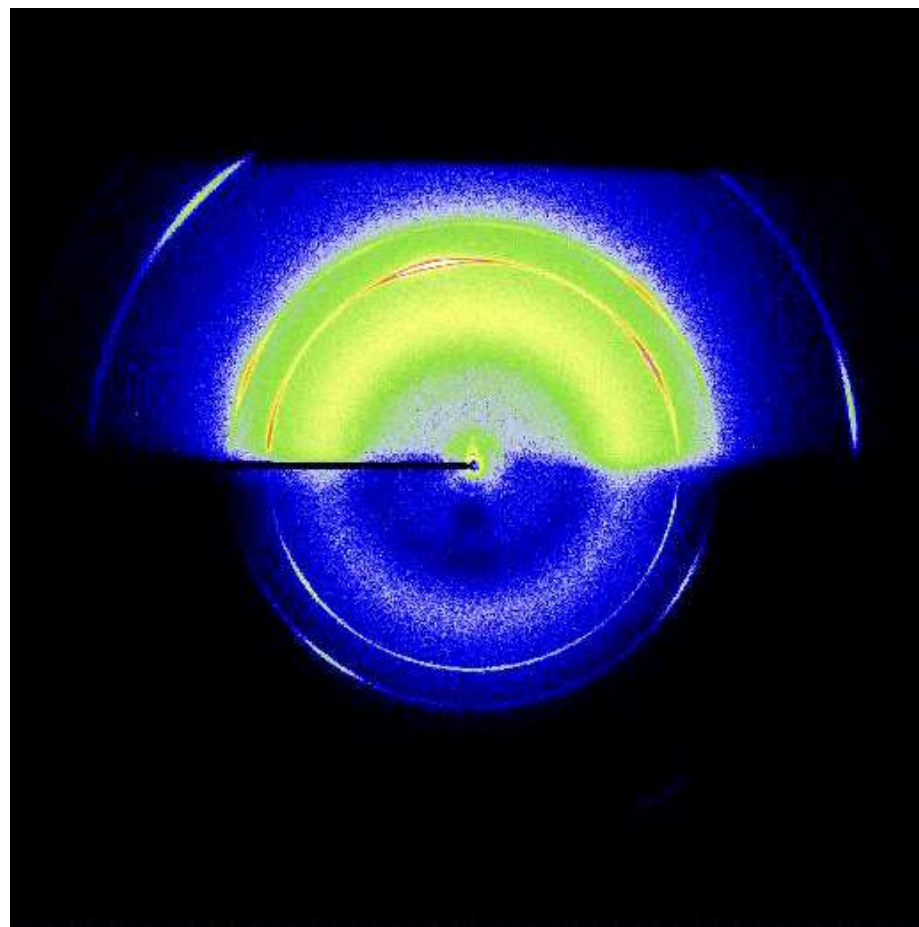
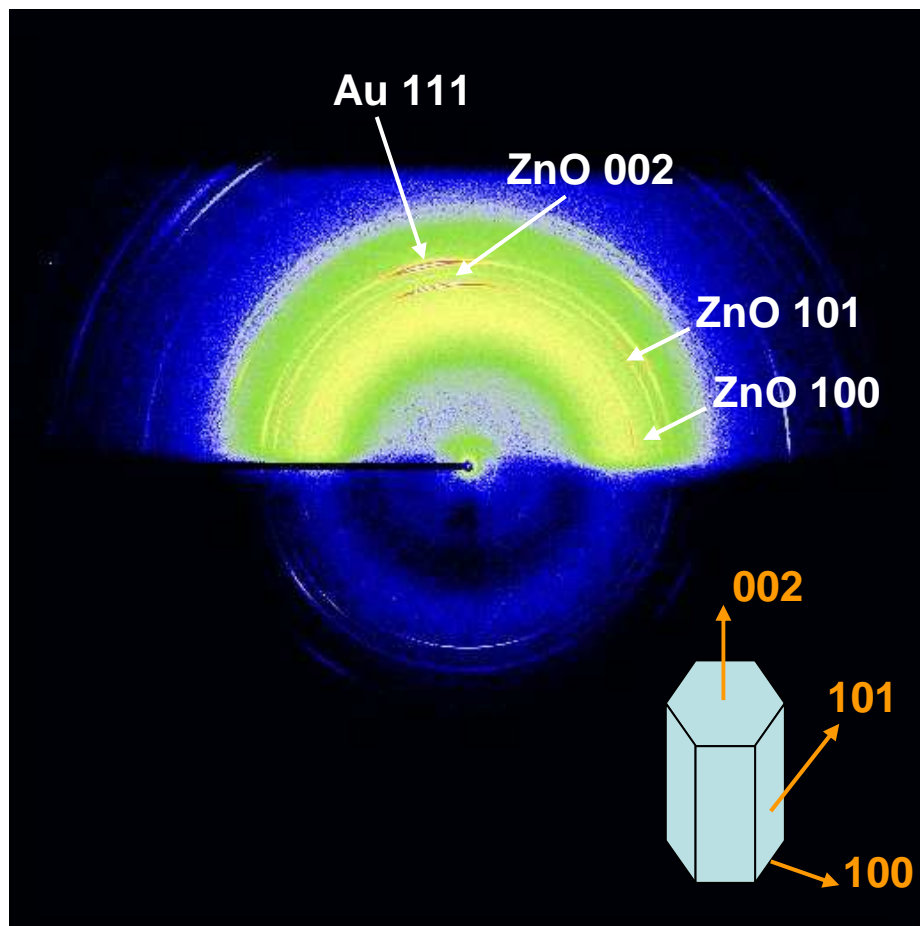


Case study: ZnO electrochemical deposition

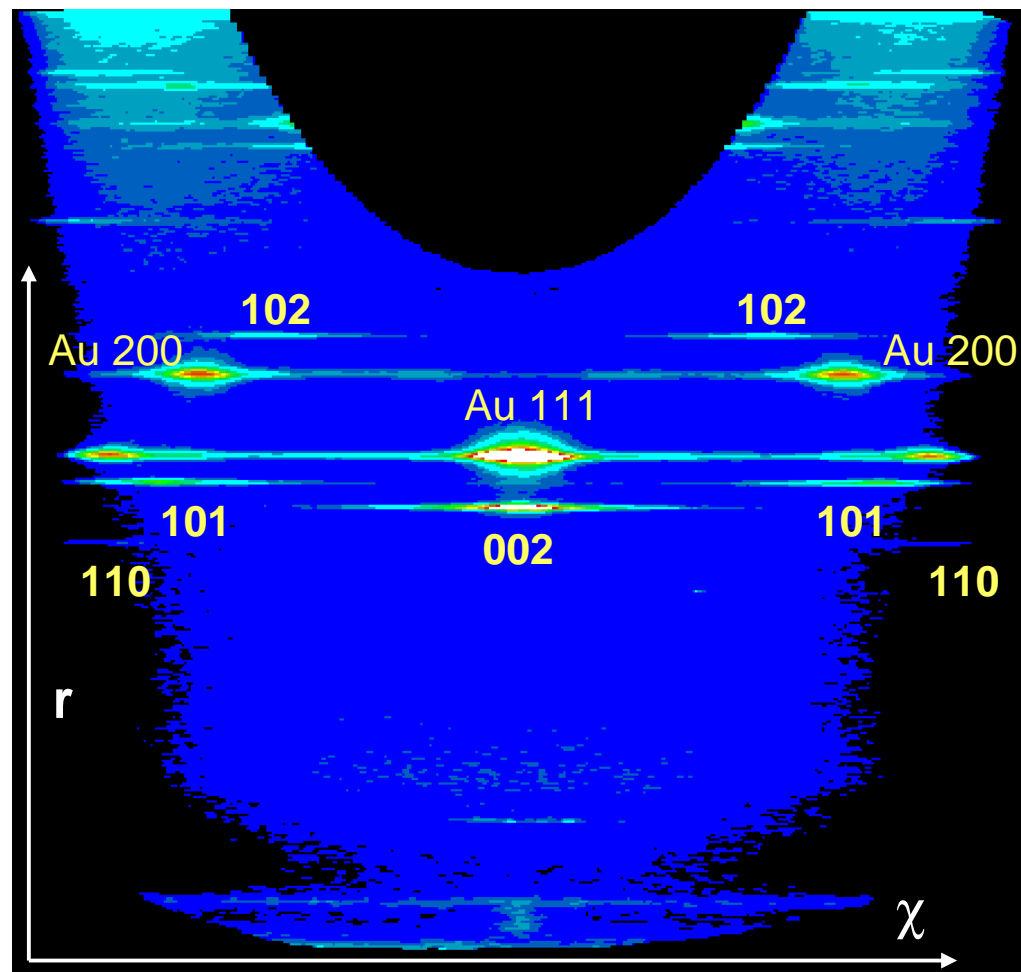
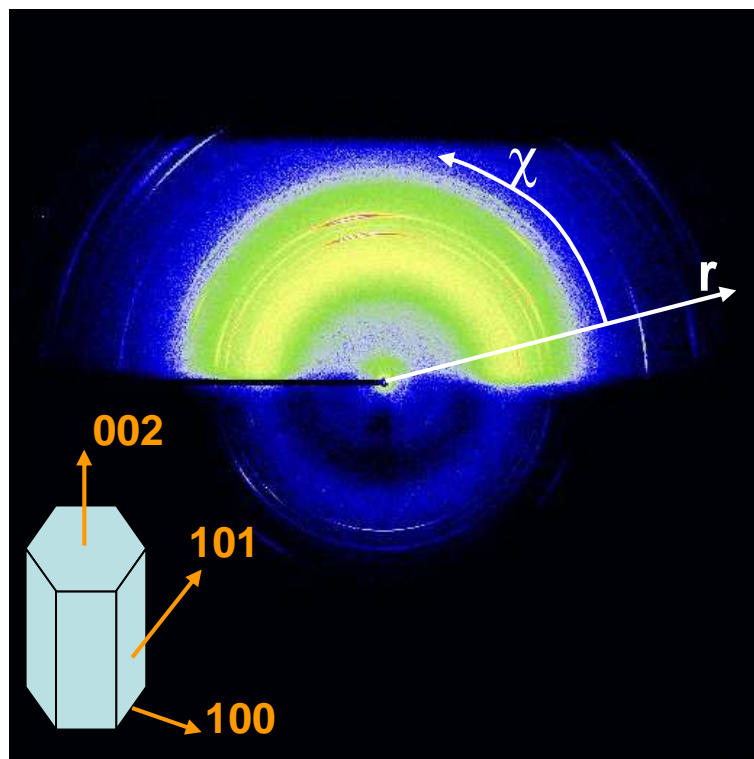


Electrochemical
cell

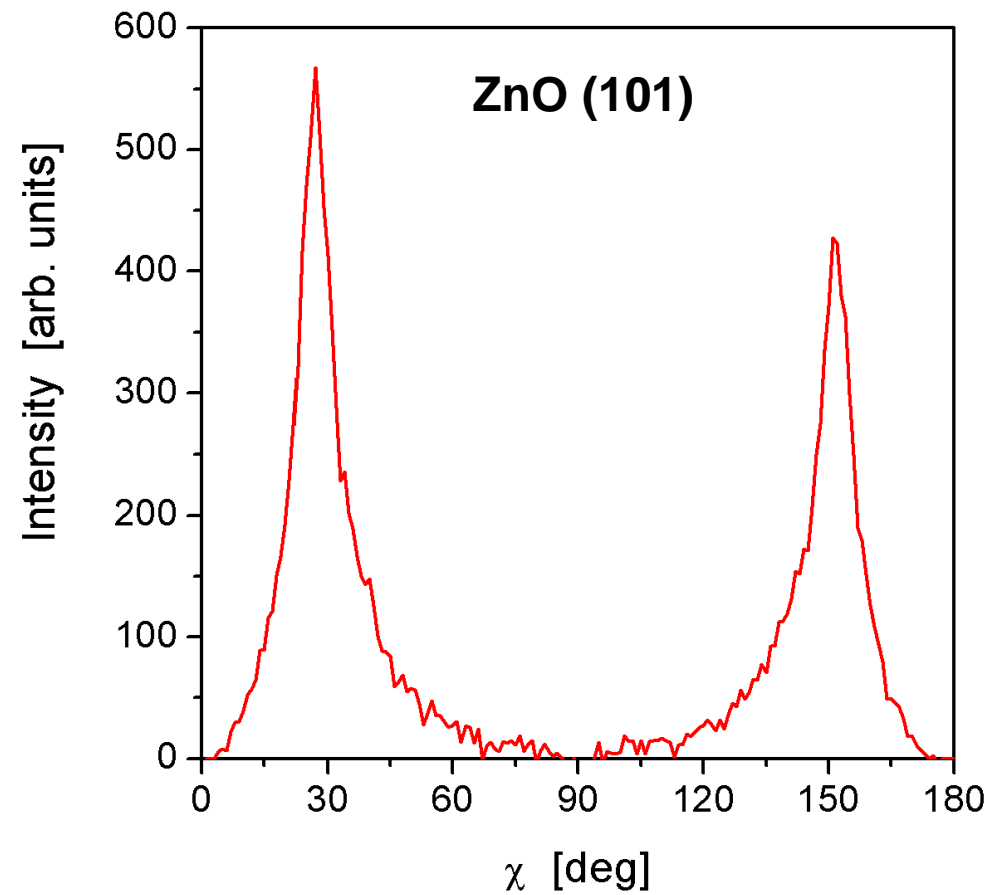
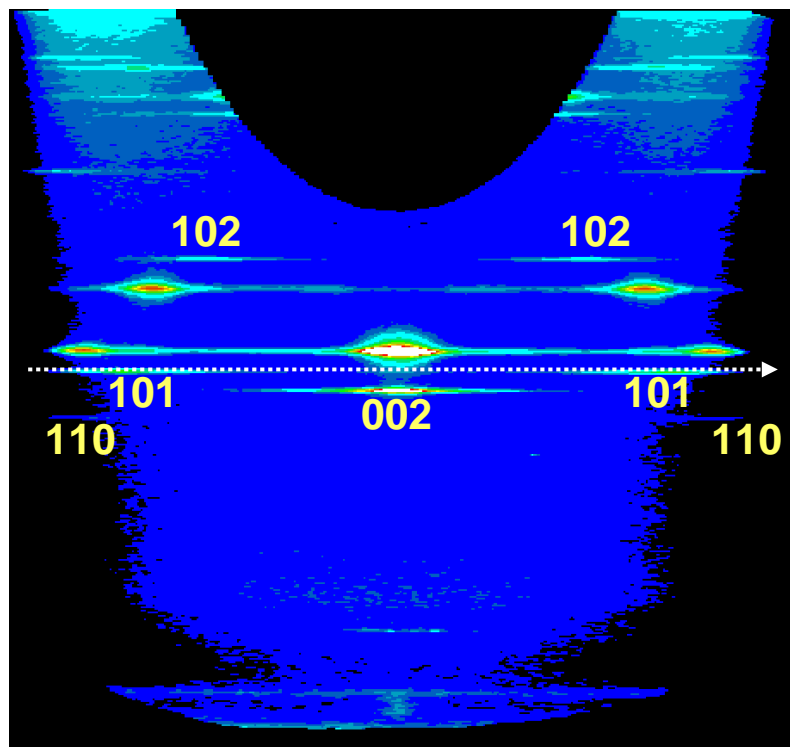
Raw data



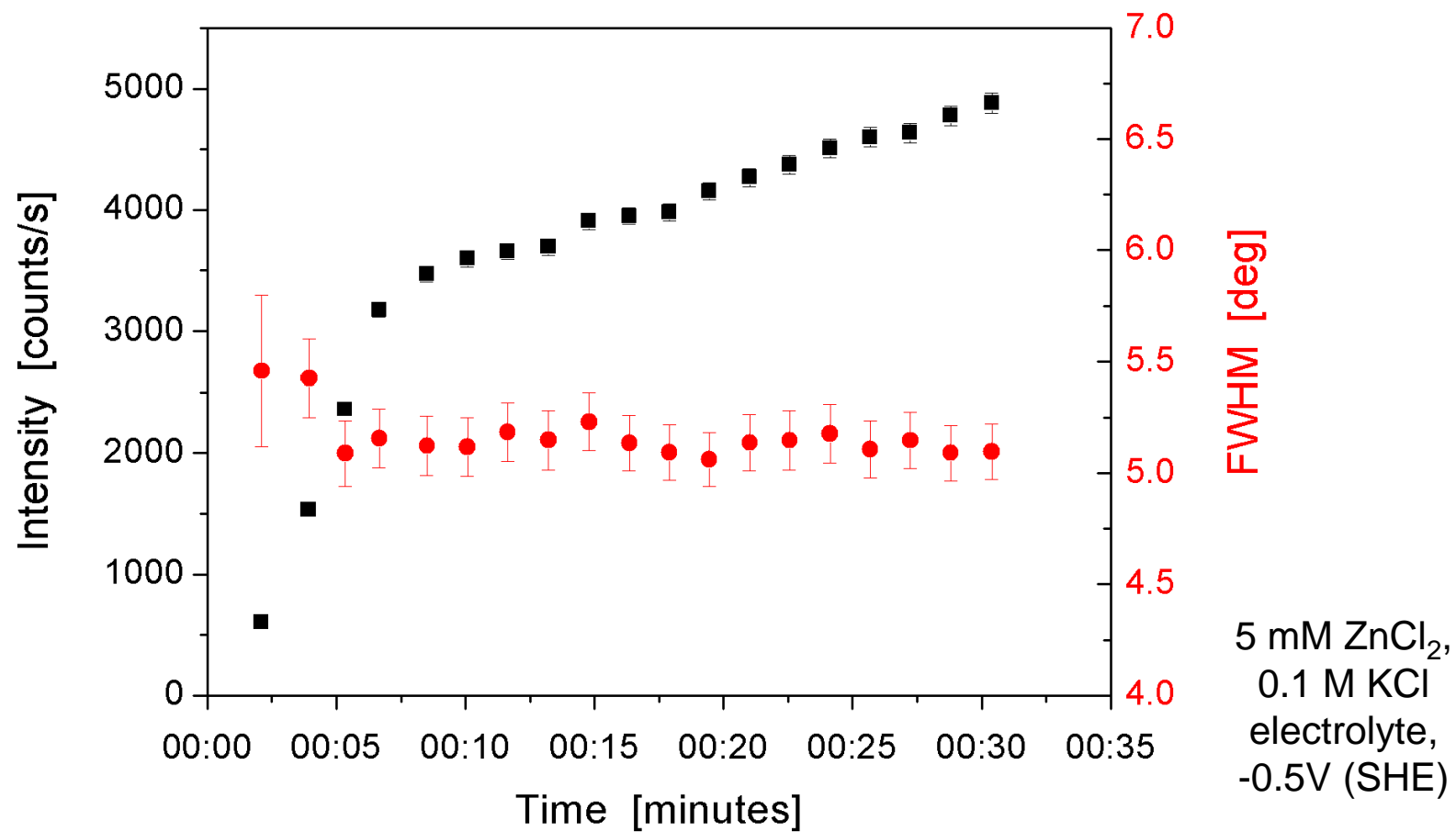
Data reduction



Data analysis



Time series results



Outline of the day

- Introduction
- Basic X-ray diffraction
 - Powders, thin films, single crystals
- Small-angle X-ray scattering
- X-ray absorption spectroscopy
 - Atomic structure
 - Electronic structure (Ben Ruck)
- Advanced X-ray diffraction
 - Anomalous diffraction
 - Total X-ray scattering
- Designing experiments
 - In situ measurements